

1 Material and Morphology Parameter Sensitivity Analysis in Particulate
2 Composite Materials

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5 **Abstract**

6 This manuscript presents a novel parameter sensitivity analysis framework for damage
7 and failure modeling of particulate composite materials subjected to dynamic loading. The
8 proposed framework employs global sensitivity analysis (GSA) to study the variance in the
9 failure response as a function of model parameters. In view of the computational complexity
10 of performing thousands of detailed microstructural simulations to characterize sensitivities,
11 Gaussian Process (GP) surrogate modeling is incorporated into the framework. In order to
12 capture the discontinuity in response surfaces, the GP models are integrated with a support
13 vector machine (SVM) classification algorithm that identifies the discontinuities within re-
14 sponse surfaces. The proposed framework is employed to quantify variability and sensitivities
15 in the failure response of polymer bonded particulate energetic materials under dynamic loads
16 to material properties and morphological parameters that define the material microstructure.
17 Particular emphasis is placed on the identification of sensitivity to interfaces between the poly-
18 mer binder and the energetic particles. The proposed framework has been demonstrated to
19 identify the most consequential material and morphological parameters under vibrational and
20 impact loads.

21 *Keywords:* Energetic materials interfaces, Surrogate model, Global Sensitivity Analysis

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1 Introduction

Dynamic behavior of particulate composites such as energetic materials subjected to high amplitude, transient (e.g., harmonic or shock) loads is often complicated due to interacting chemo-thermo-mechanical processes acting at multiple time and length scales. A significant body of work exists on understanding and characterization of the roles of various mechanisms (e.g., pore collapse, intergranular friction, particle fracture, interface debonding, dislocation pile-ups [3, 13, 59]) to the overall composite performance. The corresponding mechanistic models that describe the constituent (e.g., binder, particle, interfaces, particle-particle interactions) behavior reflect the degree of complexity of these mechanisms [60]. In mesoscale simulations with tremendously complex morphologies and material behavior, characterization of the relative roles of the competing and interacting deformation and failure mechanisms is a significant challenge. This manuscript describes a computational sensitivity analysis framework to quantify the parametric and morphological sensitivity of particulate composites subjected to dynamic loading.

A number of sensitivity analysis techniques including One-At-a-Time (OAT) [43], screening methods [31], differential analysis/local methods [38], scatter plot [32], regression analysis [5], global sensitivity analysis [45], among others [6, 44] have been developed in the past several decades (see e.g., [44] for an overview). Among these methods, OAT, differential analysis, and regression analysis have been previously used to capture parameter sensitivity in particulate composites. In energetic materials, sensitivity analyses have been largely performed using the OAT method, which probes the vicinity of a calibrated parameter set by varying one parameter at a time while other parameters are kept constant. Czerski and Proud [9] applied the OAT method on cyclotrimethylene-trinitramine to investigate its shock sensitivity to particle size and particle morphologies. Barua et al. [2] investigated the ignition sensitivity of granular explosives (GXs) and polymer-bonded explosives (PBXs) to morphology (grain volume fraction, grain size distribution) and impact velocity. One main drawback of the OAT method is the inability of accounting for the interactions between parameters since only one parameter is probed each time [43].

Differential analysis relies on (typically numerical) differentiation of the response function with respect to each input parameter [22]. For instance, Rohan and Miara [40] derived analytical sensitivity formulae for the homogenized properties of particle reinforced piezoelectric composites with respect to parameters that control particle morphology. The resulting sensitivities obtained in differential analysis are local, and the overall sensitivity of a response function to an input variable is not readily available. Furthermore, this approach is typically applicable and feasible for simple and smooth response functions, in which differentiation is possible. Numerical approaches to differentiation are often computationally costly and limited

58 to obtaining local sensitivities [19]. Regression analysis has also been employed as a sensitivity
59 analysis method [5, 55] including for particulate composites. For instance, Seuntjens et al.
60 [48] calculated a Partial Rank Correlation Coefficient (PRCC), a rank regression parameter,
61 to rank the sensitivities of physical and chemical properties that affect transport in porous
62 particulate media. Regression analysis relies on an assumed relationship between inputs and
63 outputs, which makes the obtained results dependent upon the assumed functional form [6].

64 Global sensitivity analysis (GSA) methods offer the capability to exploit the entire param-
65 eter space, and are able to probe beyond small local subspaces. For instance, Yu et al. [64]
66 employed a GSA-based high dimensional model representation (HDMR) method to assess the
67 sensitivity of mesoscale parameters on the elastic properties of a particulate composite. The
68 capability of exploring the entire parameter space is one of the main reasons that GSA has
69 a better chance to avoid misclassification of a highly sensitive parameter as non-influential
70 (or vice-versa). GSA methods also account for the interaction terms between input paramete-
71 rs [45].

72 In this manuscript, a computational sensitivity analysis framework is developed to quantify
73 the material and morphological sensitivities in the dynamic response of particulate compos-
74 ites. The proposed analysis framework is based on global sensitivity analysis (GSA) coupled
75 with Gaussian Process (or Kriging) Surrogate Modeling for computational efficiency. The pro-
76 posed framework is verified in the context of the dynamic response of Hydroxyl-terminated
77 polybutadiene (HTPB) polymeric binder reinforced with ammonium perchlorate (AP) par-
78 ticles. Analyses were performed to quantify sensitivities of material (particularly interfacial)
79 and morphological parameters on the AP-HTPB composites subjected to impact and harmonic
80 loads.

81 The execution of GSA requires a substantial number of forward simulations, particularly for
82 highly varying response functions. The nonlinear dynamic simulations of particulate composite
83 mesostructures [66] are typically extremely costly from the computational perspective. In order
84 to perform sensitivity analysis in a computationally efficient manner, the present manuscript
85 relies on surrogate (i.e., Gaussian Process) modeling of the dynamic thermo-mechanical be-
86 havior of the material mesostructure. Gaussian Process (GP) models employing commonly
87 used squared exponential kernel or rational quadratic kernels are well-known to exhibit signifi-
88 cant errors when the response function has discontinuities [52]. A novel contribution of this
89 manuscript is the development of a piece-wise continuous GP model to represent discontinuous
90 response functions. The piece-wise continuous GP model is built by employing the support
91 vector machine (SVM) - a classification algorithm [7]. Instead of using the representative
92 response surface generated by global surrogate model trained over the entire input domain,

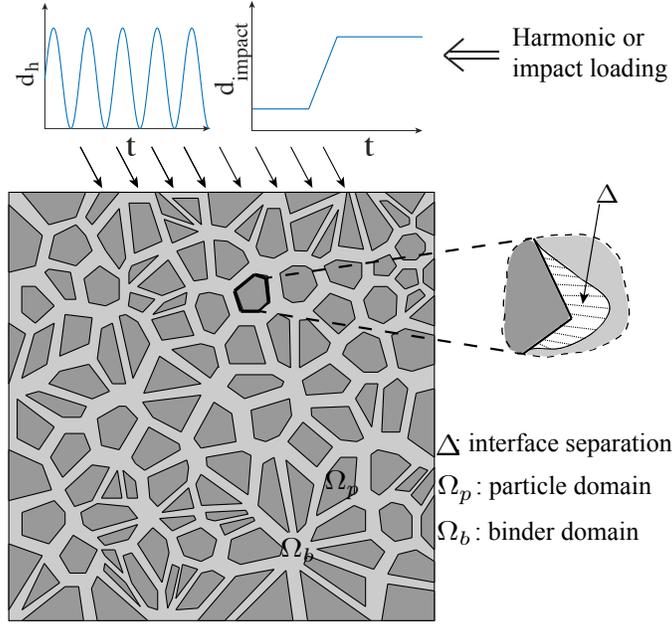


Figure 1: Problem setting: mesoscale geometry and loading.

93 several local surrogate models are generated for different parameter subdomains to capture
 94 the response surface, and SVM is utilized to identify the parameter subdomains.

95 The remainder of this manuscript is structured as follows: Section 2 provides the problem
 96 setting and the material models employed in the forward simulation of the dynamic response
 97 of particulate composites at the mesoscale. Section 3 describes of the proposed sensitivity
 98 analysis framework, including the piece-wise continuous GP surrogate model, the support
 99 vector machine classifier and the global sensitivity analysis method. Numerical examples
 100 that utilize the proposed framework to quantify the sensitivity of material properties and
 101 mesoscale geometry parameters are discussed in Section 4. Section 5 provides a summary and
 102 the conclusions.

103 2 Meso-scale Modeling of the AP-HTPB system

104 We are interested in computing the sensitivity of the dynamic response of a mesoscale repre-
 105 sentative volume to material parameters of the composite constituents as well as the mesoscale
 106 morphology. Consider a representative volume that consists of randomly positioned particles
 107 with varying particle sizes and shapes as illustrated in Fig. 1. Under the applied loading, the
 108 composite is taken to undergo dissipative, large thermo-mechanical deformations. In order
 109 to limit the complexity of the physics involved, the amplitude of the loads are taken small
 110 enough and possible chemical reactions are excluded. The interfaces between the particles
 111 and the binder are allowed to progressively degrade and debond. In view of the short loading

112 times, the thermal process within the representative volume is assumed to be adiabatic. The
 113 particles are taken to exhibit elastic behavior under the applied loading. The constitutive
 114 models of the binder and the interfaces are described below.

115 2.1 Binder Constitutive Model

116 The binder is taken to be significantly softer than the particles, and assumed to exhibit vis-
 117 coelastic response [20, 62]. The Cauchy stress within the binder is expressed in terms of the
 118 hereditary integral as a function of the shear relaxation modulus as

$$\boldsymbol{\sigma} = \int_0^t 2G(t, \tau) \mathbf{D}'(\tau) d\tau + K \frac{\ln J}{J} \boldsymbol{\delta} \quad (1)$$

119 where, $\mathbf{D}'(\tau)$ is the deviatoric component of the deformation-rate tensor, $\mathbf{D} = (\mathbf{L}^T + \mathbf{L})/2$,
 120 with \mathbf{L} the velocity gradient tensor ($\mathbf{L} = \dot{\mathbf{F}}\mathbf{F}^{-1}$ with \mathbf{F} the deformation gradient tensor),
 121 and $G(t, \tau)$ the shear relaxation modulus. The volumetric part of the deformation is taken
 122 to behave elastically and K is the bulk modulus. J is the determinant of the deformation
 123 gradient, and $\boldsymbol{\delta}$ is the Kronecker delta.

124 The shear relaxation modulus, $G(t, \tau)$, is idealized using the Prony Series representation:

$$G(t, \tau) = \frac{T(\tau)}{T_{\text{ref}}} G_{h\infty} \left[1 + \sum_{i=1}^n p_i \exp\left(\frac{-(\xi(t) - \xi(\tau))}{q_i}\right) \right] \quad (2)$$

125 where, $G_{h\infty}$ is the steady-state shear modulus of the binder, T the temperature, T_{ref} the
 126 reference temperature, p_i and q_i are the fitting parameters of the n pairs of Prony Series
 127 prescribing the relative modulus, the relaxation time of the i_{th} Prony Series term, respectively,
 128 and:

$$\xi(t) = \int_0^t \frac{1}{a(T(\tau))} d\tau \quad (3)$$

129 where, $a(T)$ is the Williams-Landel-Ferry (WLF) empirical time-temperature shift function:

$$\log(a(T)) = \frac{A(T - T_{\text{ref}})}{B + T - T_{\text{ref}}} \quad (4)$$

130 A and B are material parameters. In view of the low conductivity and short loading time, the
 131 thermal process within the representative volume is assumed to be adiabatic. The viscoelastic
 132 dissipation induced heating under the adiabatic assumption is given by:

$$\dot{T} = \frac{1}{C_{V_b} \dot{W}_d} = \frac{2G_{h\infty}}{C_{V_b}} \frac{T(t)}{T_{\text{ref}}} \sum_{i=1}^n \frac{p_i}{q_i} \boldsymbol{\varepsilon}_d^i(t) : \boldsymbol{\varepsilon}_d^i(t) \quad (5)$$

133 where W_d is the dissipated work per unit reference volume, C_{V_b} the heat capacity per reference
 134 volume, and

$$\varepsilon_d^i(t) = \int_0^t \exp\left(\frac{-(\xi(t) - \xi(\tau))}{q_i}\right) \mathbf{D}'(\tau) d\tau \quad (6)$$

135 2.2 Interface Constitutive Model

136 Prior experimental studies (e.g., [39, 65]) demonstrated that debonding at particle/binder
 137 interfaces is an important failure mechanism that may result in localized hot-spots in the
 138 context of energetic composites. Progressive damage accumulation and interface fracture is
 139 modeled by employing Cohesive Zone Modeling (CZM). CZM is well-known to accurately
 140 describe fracture processes particularly when the fracture path is pre-defined such as in the
 141 case of interface debonding problems [12].

142 In this manuscript, a bilinear traction-separation law is employed in the context of CZM
 143 following Ref. [20]. Let $\lambda(\mathbf{x})$ denote a normalized equivalent displacement jump measure at
 144 an interface point, $\mathbf{x} \in S$ defined as [57]:

$$\lambda = \begin{cases} \sqrt{\left(\frac{\Delta_n}{\Delta_{cn}}\right)^2 + \left(\frac{\Delta_t}{\Delta_{ct}}\right)^2}, & \Delta_n \geq 0 \\ \frac{|\Delta_t|}{\Delta_{ct}}, & \Delta_n < 0 \end{cases} \quad (7)$$

145 where Δ_n and Δ_t are the normal and tangential components of the separation vector; and
 146 Δ_{cn} and Δ_{ct} are the critical normal and critical tangential separations under pure mode I and
 147 mode II conditions, respectively that result in traction free surface. Let λ_{ul} denote a history
 148 variable describing the peak normalized separation measure throughout the loading history:

$$\lambda_{ul} = \max_{\tau \in [0, t]} \lambda(\tau) \quad (8)$$

149 The damage state, d_m , at a material point is then expressed as a function of the history
 150 variable:

$$d_m = \begin{cases} 0, & 0 \leq \lambda_{ul} \leq \eta_0 \\ \frac{\lambda_{ul} - \eta_0}{(1 - \eta_0)\lambda_{ul}}, & \eta_0 < \lambda_{ul} \leq 1 \\ 1, & \lambda > 1 \end{cases} \quad (9)$$

151 in which, η_0 is the elastic separation limit. The traction-separation relationship is expressed
 152 as a function of damage, d_m , as:

$$t_{nc} = (1 - d_m) \frac{T_{\max}}{\eta_0} \frac{\Delta_n}{\Delta_{cn}} \quad (10)$$

$$t_{tc} = (1 - d_m) \frac{T_{\max}}{\eta_0} \frac{\Delta_t}{\Delta_{ct}} \quad (11)$$

154 where, t_{nc} and t_{tc} are normal and tangential traction components; T_{\max} denotes the maximum
 155 interfacial traction.

156 The interpenetration between binder and particle is prevented by using the penalty contact
 157 algorithm. A linear relation between interpenetration ($|\Delta_n|$) and penalty traction p_n is:

$$p_n = \beta_p |\Delta_n| H(-\Delta_n) \quad (12)$$

158 where β_p is the penalty parameter; and H the Heaviside function. A regularized Coulomb law
 159 is employed to model the post-debonding frictional behavior along the particle-binder interface:

$$t_{tf} = \mu \frac{\dot{\Delta}_t}{\sqrt{\dot{\Delta}_t^2 + (\epsilon)^2}} |p_n| \quad (13)$$

160 where ϵ is a regularization parameter, and as it approaches zero, Eq. 13 recovers the classical
 161 Coulomb law. The superposed dot indicates material time derivative. μ is the friction coefficient
 162 that increases with accumulating damage:

$$\mu = d_m \mu_0 \quad (14)$$

163 The overall local normal and tangential tractions considering the cohesive and the contact
 164 behavior are expressed as:

$$\mathbf{t}_n = [H(\Delta_n)t_{nc} + (H(\Delta_n) - 1)p_n]\hat{\mathbf{n}} \quad (15)$$

$$\mathbf{t}_t = [t_{tc} + (1 - H(\Delta_n))t_{tf}]\hat{\mathbf{t}} \quad (16)$$

166 where $\hat{\mathbf{n}}$ and $\hat{\mathbf{t}}$ are the normal and tangential unit vectors at the interface point. The decohesion
 167 process is assumed to generate negligible heat compared to the interface frictional heat. The
 168 total dissipated energy rate at the interface is assumed to be due to friction alone, which in
 169 turn leads to adiabatic temperature change as:

$$\dot{T}_{\text{int}} = \frac{t_{tf}\dot{\Delta}_t}{C_{V_{\text{int}}}} \quad (17)$$

170 where $C_{V_{\text{int}}}$ is the specific heat capacity at the interface.

171 2.3 Mesoscale Geometry

172 The morphology of the particulate composite mesostructure significantly affects the failure and
 173 initiation behavior in energetic materials [23, 24, 30, 58]. Despite the recognition of morphology
 174 as a key factor influencing material dynamic behaviors, the relative importance/sensitivities of

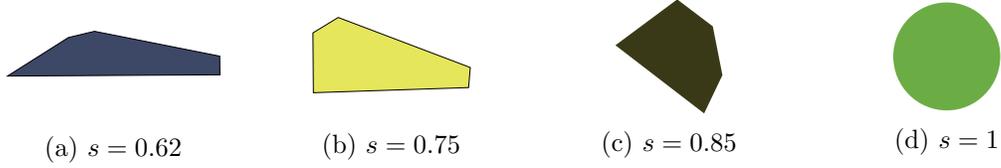


Figure 2: Shapes with different sphericity.

175 morphological parameters with respect to the behavior of interest remains to be systematically
 176 investigated and quantified. In the current study, the mesoscale morphology is probabilisti-
 177 cally parameterized by the following: (1) particle size distribution, (2) particle area fraction
 178 distribution, and (3) particle sphericity (i.e., particle shape) distribution.

179 The particle area fraction and size distribution significantly affect the particle/binder in-
 180 terfacial behavior and the interaction between particles [1, 34]. For instance, the burning rate
 181 tailorability of propellants and explosive materials is partially controlled by particle size dis-
 182 tributions [56]. In this study, the size of a particle is quantified directly as the area occupied
 183 by the particle.

184 Energetic particles vary significantly in shape and complexity which directly influence the
 185 particle-particle and particle-binder interactions. The impact of particle shape in localized
 186 increases in temperature has been reported in Refs. [11, 23, 30]. In the current manuscript,
 187 particle shapes are idealized as polygons, and the shape distribution is defined by the distri-
 188 bution of the parameter “sphericity”. The sphericity is defined as the ratio of the perimeter
 189 of a sphere with equivalent particle size to perimeter of the polyhedron:

$$s = \frac{P_{eq}}{L_{edge}} \quad (18)$$

190 where $P_{eq} = 2\sqrt{\pi A}$, with A the particle size, L_{edge} the perimeter of the polyhedron. The
 191 sphericity value for each particle varies in the range $(0, 1)$. Sphericity near zero represents
 192 a thin and long shape, while near unit sphericity represents a shape that is very similar to
 193 a circle. Different shapes controlled by different sphericities are illustrated in Fig. 2. In the
 194 context of numerical verifications, the sphericity distribution within the particulate composite
 195 is taken to be Gaussian:

$$f(s) = N(\mu_s, \sigma_s) \quad (19)$$

196 where, μ_s and σ_s are the sphericity expectation and variance, respectively.

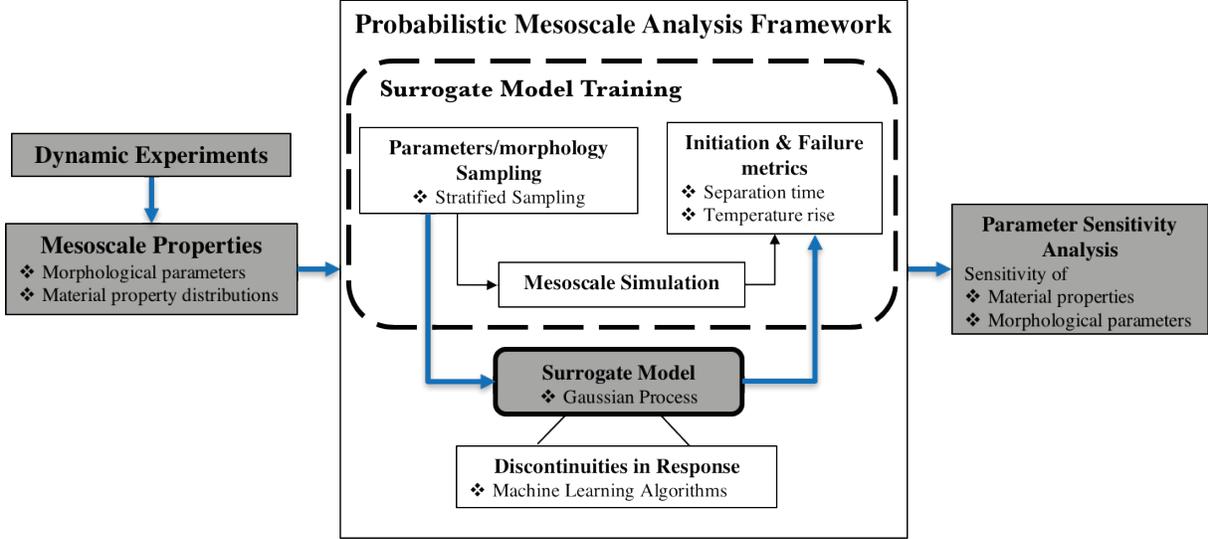


Figure 3: Overview of the proposed sensitivity analysis framework.

3 Sensitivity Analysis Framework

The proposed framework is established based the idea that parameter sensitivities of particulate materials should be quantified in a global sense (i.e., across the entire parameter space) to understand the role that each parameter plays on the corresponding failure mechanisms, and in the presence of discontinuous response surfaces under the applied loading. The proposed framework is schematically illustrated in Fig. 3.

The proposed framework relies on a parametric description of the mesoscale morphology as well as the constitutive response of the constituents (i.e., inclusion, binder, interface). The morphological and constitutive parameters are taken to be random variables. In order to perform sensitivity analysis, a set of samples is drawn to cover the entire subspace of physically plausible parameters. In order to avoid the necessity of performing direct mesoscale simulations for each sample point within the set, the proposed framework relies on adequately trained surrogate modeling (Gaussian Process, or GP) that relates the parameters to the response surface of a chosen mesoscale metric (e.g., interface separation, temperature rise, initiation). The sensitivity assessment is based on the Global Sensitivity Analysis (GSA) method. In GSA, the variance in the model output (i.e., mesoscale metric) results from the variance in the model input parameters (i.e., material parameters and the morphology), as well as the variance induced by the model input interactions. In this study, the contributions of the variance in each input to the variance in the output is characterized using the first order sensitivity and total effect indices. First order index represents the contribution to the output variance from the parameter itself, while total effect index also includes the interactions between the parameter and the other parameters.

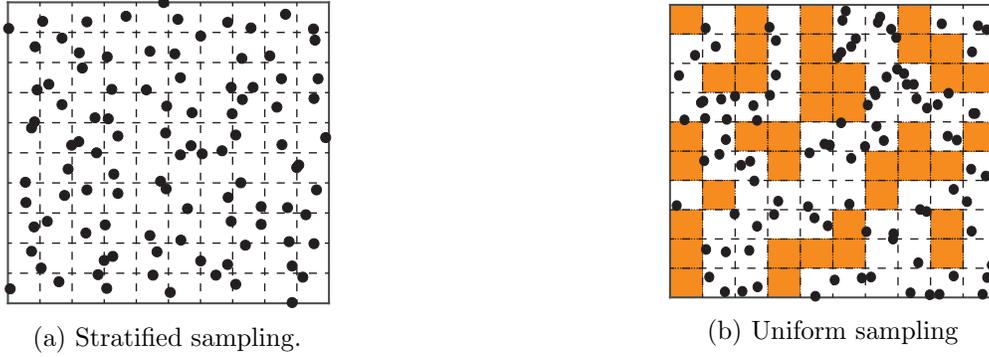


Figure 4: Comparison between stratified sampling and uniform sampling.

3.1 Stratified Sampling

The sampling of the parameter space is performed such that the following two requirements are satisfied: (1) The samples must cover the entire parameter space. This is in view of the fact that we seek to compute the sensitivity of the parameters across the entire parameter space, in contrast to local sensitivities. (2) The total number of samples must be as small as possible for computational efficiency, in view of the computational complexity of the mesoscale simulations. In the current study, the Stratified Sampling method [33] is employed. Stratified sampling ensures certain subspaces are not overrepresented or underrepresented. The idea of stratified sampling is to partition the parameter space to multiple non-overlapping subspaces, and to ensure that a set of samples from each subspace is drawn. Sampling within each subspace allows probing the local characteristics of the response function throughout the parameter space.

The benefit of employing the stratified sampling method is graphically illustrated in Fig. 4. In Fig. 4a, stratified sampling is applied to generate 100 randomly distributed samples in a two dimensional parameter space. The entire region is evenly divided as $10 \times 10 = 100$ subspaces with one sample per subspace. In Fig. 4b, uniform sampling was employed to draw 100 samples. The uniform sampling method fails to generate samples in 38 of the subspaces probed by stratified sampling, and fails to sample from large swaths of the parameter space. Alternative sampling approaches such as Sobol sequences [50] and other quasi Monte Carlo methods that uniformly sample the entire parameter space could also be employed with similar efficiency compared with the stratified sampling.

3.2 Gaussian Process Model

In view of the significant computational cost of performing mesoscale dynamic simulations, probing the entire parameter space within the context of sensitivity analysis through mesoscale simulations as a forward solver is computationally prohibitive. In this study, we employ Gaus-

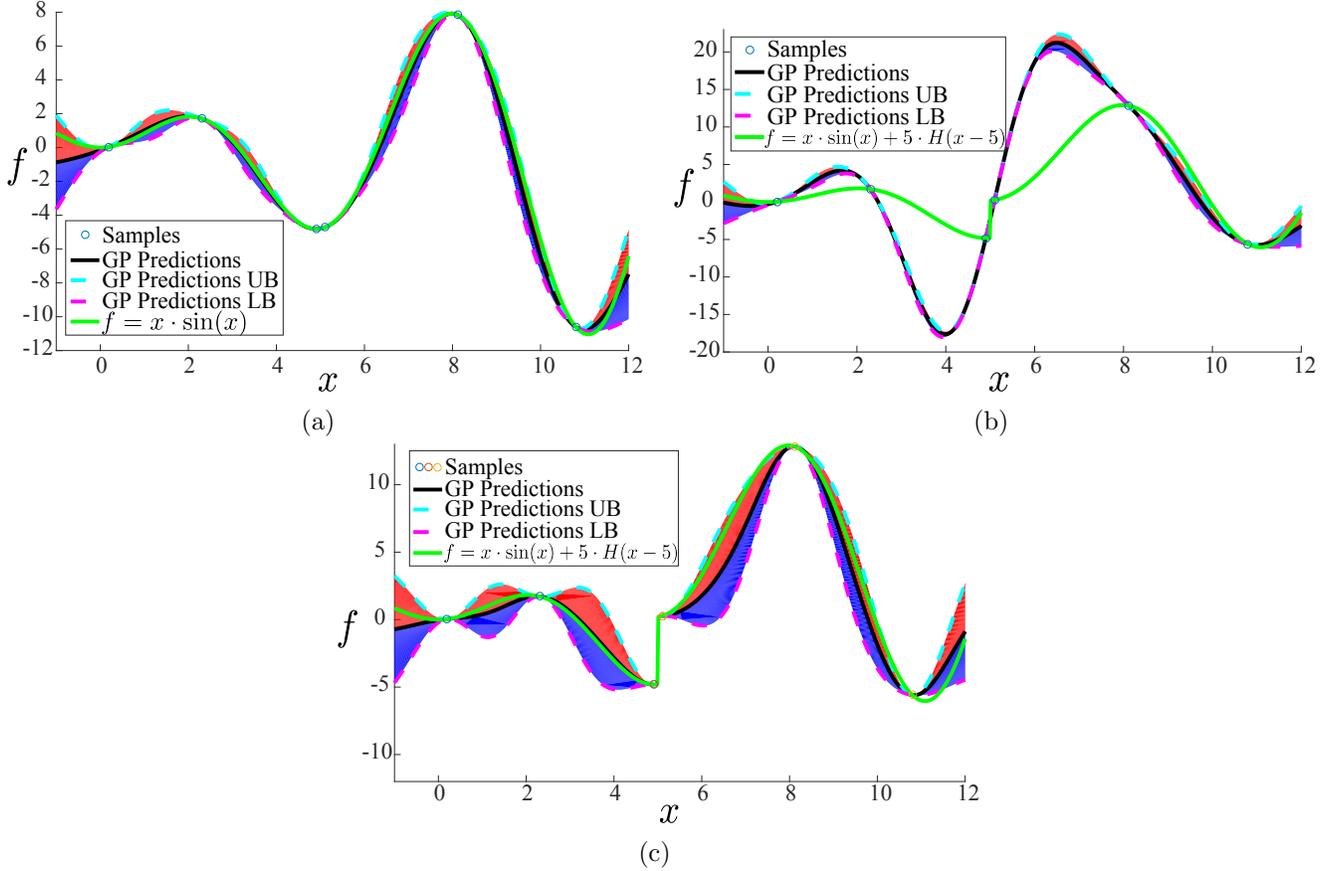


Figure 5: Example of GP model and discontinuity. Illustration of (a) a GP surrogate of function, $f = x \cdot \sin(x)$ trained with 6 training points; (b) a GP surrogate of function, $f = x \cdot \sin(x) + 5 \cdot H(x - 5)$ trained with 6 samples; (c) two local GP surrogates of function, $f = x \cdot \sin(x) + 5 \cdot H(x - 5)$ trained with 6 samples.

244 sian Process modeling as the surrogate forward simulator. In this approach, a small suite of
 245 mesoscale simulations are performed with randomly sampled parameters to generate discrete
 246 values of the response function, called training points. The training points are employed to con-
 247 struct the surrogate model, which is in turn employed to predict the response with randomly
 248 sampled prediction points.

249 Gaussian process (or Kriging) is a method of interpolation for which the interpolated
 250 values are modeled by a Gaussian process governed by prior covariance. The Gaussian process
 251 method is probabilistic, and it generates a distribution instead of a single prediction value at
 252 the prediction point. The variance of the prediction distribution is related to the proximity
 253 of the prediction point input (i.e., the parameter set that correspond to the prediction point)
 254 and the training point inputs as illustrated in Figure 5a.

255

The GP model is expressed in the following form:

$$G_p(\mathbf{u}) = \mathbf{h}(\mathbf{u})^T \cdot \boldsymbol{\beta} + Z(\mathbf{u}) \quad (20)$$

256

where $\mathbf{u} = (u^1, u^2, \dots, u^d)$ denotes the vector of input parameters, each of which is a random variable, d denotes the number of parameters (i.e., the dimensionality of the problem), \mathbf{h} is the trend of the model, $\boldsymbol{\beta}$ is the vector of trend coefficients, and Z is a stationary Gaussian process with zero mean. The trend of the model is taken to be up to second order polynomial regression model in this study, and higher order polynomial trend could be achieved via increasing the number of training samples.

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The covariance between sample outputs is defined as

$$Cov[Z(\mathbf{a}), Z(\mathbf{b})] = \sigma_z^2 \cdot R(\mathbf{a}, \mathbf{b}) \quad (21)$$

263

where σ_z^2 is the process variance and $R(\cdot, \cdot)$ is the correlation function between two input points \mathbf{a} and \mathbf{b} . Squared-exponential function, a commonly used correlation function is employed:

264

$$R(\mathbf{a}, \mathbf{b}) = \exp \left[- \sum_{i=1}^d \theta_i (a_i - b_i)^2 \right] \quad (22)$$

265

where $\theta_i \in \boldsymbol{\theta} = (\theta_1, \dots, \theta_d)$ is a parameter that indicates the correlation between the points within dimension i , a_i and b_i are the i_{th} components of \mathbf{a} and \mathbf{b} , respectively.

266

267

The expectation value and variance of the GP model prediction distribution at input point \mathbf{u} are expressed as:

268

$$\mu_G(\mathbf{u}) = \mathbf{h}(\mathbf{u}) \cdot \boldsymbol{\beta} + \mathbf{r}(\mathbf{u})^T \mathbf{R}^{-1}(\mathbf{g} - \mathbf{F}\boldsymbol{\beta}) \quad (23)$$

269

$$\sigma_G^2(\mathbf{u}) = \sigma_z^2(\mathbf{u}) - \begin{bmatrix} \mathbf{h}(\mathbf{u})^T & \mathbf{r}(\mathbf{u})^T \end{bmatrix} \begin{bmatrix} \mathbf{0} & \mathbf{F}^T \\ \mathbf{F} & \mathbf{R} \end{bmatrix}^{-1} \begin{bmatrix} \mathbf{h}(\mathbf{u}) \\ \mathbf{r}(\mathbf{u}) \end{bmatrix} \quad (24)$$

270

where $\mathbf{r}(\mathbf{u})$ is a vector containing the covariance between \mathbf{u} and each of the n training points $(\mathbf{u}_1, \dots, \mathbf{u}_n)$, \mathbf{R} is $n \times n$ matrix containing the correlation between each pair of training points, \mathbf{g} is the vector of response outputs at each of the training points, and \mathbf{F} is $n \times q$ matrix with rows $\mathbf{h}(\mathbf{u}_i)^T$, where q is the number of trend function terms for each row.

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The process variance σ_z^2 and character parameter $\boldsymbol{\theta}$ are determined through maximum likelihood estimation. Taking the log of the probability of observing the response values \mathbf{g} given the covariance matrix \mathbf{R} :

275

276

$$\log[p(\mathbf{g}|\mathbf{R})] = -\frac{1}{n} \log |\mathbf{R}| - \log(\hat{\sigma}_z^2) \quad (25)$$

277

where $|\mathbf{R}|$ indicates the determinant of \mathbf{R} , and $\hat{\sigma}_z^2$ is the optimal value of the variance given

278 an estimate of $\boldsymbol{\theta}$ and is defined by

$$\hat{\sigma}_z^2 = -\frac{1}{n}(\mathbf{g} - \mathbf{F}\boldsymbol{\beta})^T \mathbf{R}^{-1}(\mathbf{g} - \mathbf{F}\boldsymbol{\beta}) \quad (26)$$

279 Maximizing Eq. 25 gives the maximum likelihood estimate of $\boldsymbol{\theta}$, which in turn defines σ_z^2 .

280 3.3 Discontinuities and Classifier

281 The response of particulate composites under dynamic loading conditions are marked by the
 282 presence of discontinuities induced by the discrete failure events that are functions of both
 283 material and morphological parameters. In contrast, the Gaussian Process models employing
 284 squared-exponential covariance functions or rational quadratic kernels exhibit significant errors
 285 when the response function is discontinuous [52]. Figure 5b illustrates the GP approximation
 286 of a function similar to that shown in Fig. 5a, but that contains a discontinuity. With the same
 287 training samples used in Fig. 5a, the GP model shown in Fig. 5b fails to accurately represent
 288 the discontinuous function.

289 In this manuscript, a piece-wise continuous GP model is proposed to account for the pres-
 290 ence of discontinuities in the response functions. The piece-wise continuous GP model is built
 291 by employing the support vector machine (SVM) - a classification algorithm to solve the model
 292 selection problem.

293 Let $\Theta \subset \mathbb{R}^d$ denote the parameter space, which yields a discontinuous response function,
 294 $y(\mathbf{u})(\mathbf{u} \in \Theta)$. The response function is assumed to be continuous within q subdomains ($\Theta_a \subset$
 295 $\Theta; a = 1, \dots, q$) separated by $(m - 1)$ dimensional hyperplanes. The corresponding piecewise
 296 continuous GP approximation of the response function is expressed as:

$$GP(\mathbf{u}) = \sum_{a=1}^q N_a(\mathbf{u})GP_a(\mathbf{u}) \quad (27)$$

297 in which GP_a denotes the continuous GP approximation valid within Θ_a ; and N_a is a piecewise
 298 constant shape function:

$$N_a(\mathbf{u}) = \begin{cases} 1, & \text{if } \mathbf{u} \in \Theta_a \\ 0, & \text{elsewhere} \end{cases} \quad (28)$$

299 Figure 5c illustrates a piece-wise continuous GP model in predicting the discontinuous
 300 function shown in Fig. 5b. The continuous parts on both sides of the discontinuity of the
 301 function are captured within the $\mu_G \pm 3\sigma_G$ interval by the two local surrogates.

The primary difficulty in the construction of the proposed surrogate model in Eq. 27 is
 that the parameter subdomains Θ_a are not known a-priori. We seek to identify the separator
 hyperplanes between the parameter subdomains based on a classification algorithm of the ma-

chine learning theory [26]. SVM is a discriminative classifier formally defined by a separating hyperplane/decision boundary [54]. For given set of input points $(\mathbf{u}_1, \dots, \mathbf{u}_n)$ with their corresponding outputs $(\mathbf{y}_1, \dots, \mathbf{y}_n)$, ϕ is a mapping function from the input space to a hypothesis and potentially infinite-dimensional feature space [47] in which the inner product is $\langle \phi(\mathbf{u}_i), \phi(\mathbf{u}_j) \rangle$ such that the training set is linearly separable. The hyperplane is found through the following optimization problem:

$$\max_{\mathbf{w}, b} \min_i \{ \mathbf{y}_i [\langle \mathbf{w}, \phi(\mathbf{u}_i) \rangle - b] \}$$

where $\langle \cdot, \cdot \rangle$ represents inner product, \mathbf{w} is a vector, and b is a real number. The expression $(\langle \mathbf{w}, \phi(\mathbf{u}_i) \rangle - b)$ is the distance between \mathbf{u}_i and the decision boundary. The sign of $\mathbf{y}_i \{ \langle \mathbf{w}, \phi(\mathbf{u}_i) \rangle - b \}$ is positive for the correct classifications and negative for the incorrect classifications. If all data sets are linearly separable, $\gamma = \min \{ \mathbf{y}_i [\langle \mathbf{w}, \phi(\mathbf{u}_i) \rangle - b] \}$ is positive. It is proven in [8] that the decision function

$$f(\mathbf{u}) = \text{sign}(\langle \mathbf{w}, \phi(\mathbf{u}) \rangle - b) \quad (29)$$

is equivalent to

$$f(\mathbf{u}) = \text{sign}\left(\sum_{i=1}^n \alpha_i \mathbf{y}_i \langle \phi(\mathbf{u}_i), \phi(\mathbf{u}) \rangle - b\right). \quad (30)$$

From this expression, it can be observed that only those points with non-zero α_i (called support vector) determine the hyperplane. The dot product between data point vectors, $\phi(\mathbf{u}_i)$ and $\phi(\mathbf{u}_j)$, is written as $K_{ij} = \langle \phi(\mathbf{u}_i), \phi(\mathbf{u}_j) \rangle$. Eq. 30 becomes:

$$f(\mathbf{u}) = \text{sign}\left(\sum_{i=1}^n \alpha_i \mathbf{y}_i \mathbf{K}(\mathbf{u}_i, \mathbf{u}_j) - b\right). \quad (31)$$

where \mathbf{K} is the kernel matrix. In current study, quadratic kernel $K(\mathbf{u}_i, \mathbf{u}_j) = (\langle \mathbf{u}_i, \mathbf{u}_j \rangle + 1)^2$ is employed.

An additional and equally important problem is the identification of the number of discontinuities, q , within the parameter space. While not straightforward, those problems that exhibit unknown number of discontinuities could be addressed through cluster analysis in data mining, such as the elbow method [14], X-means clustering [35], Akaike information criterion [42], the silhouette method [41]. In the examples in this manuscript, the number of discontinuities are dictated by the physics of the problem and assumed to be known a-prior.

The performance of the classifier is assessed by k -folder cross-validation, which is a non-exhaustive cross-validation method [25] as illustrated in Fig. 6. The full set of input parameters is separated into k sets of equal size. An arbitrary subset is taken as the *validation set*, whereas the remaining $(k - 1)$ sets are employed as *training sets*. The training sets are employed to

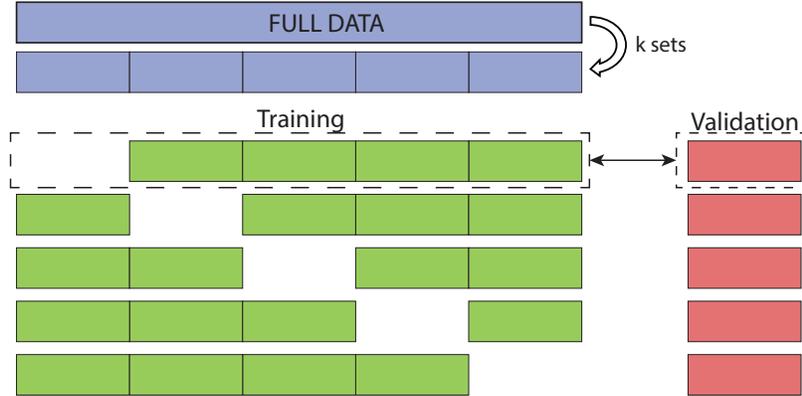


Figure 6: Cross-validation of the classification algorithm.

Table 1: Confusion matrix.

	Positive (Predict)	Negative (Predict)
Positive (Actual)	<i>True</i>	<i>False</i>
Negative (Actual)	<i>False</i>	<i>True</i>

construct the classifier. The validation set is then employed to assess whether the predicted classification (i.e., the associated parameter subdomain of each input, \mathbf{u}_i) is accurate. This process is repeated k times by assigning a different subset as the validation set.

Confusion matrix (i.e., error matrix) is produced by cross-validation to quantify the performance of a classifier (Table 1). Confusion matrix is a specific table layout that allows visualization of the performance of an algorithm. Each row of the confusion matrix represents the true class (i.e., the parameter subdomain), while each column of the confusion matrix represents the prediction of the classifier. For example, the $(i, j)_{\text{th}}$ element in the confusion matrix indicates the number or percentage of cases which belong to i_{th} parameter subdomain while predicted as j_{th} parameter subdomain. The diagonal terms represent the number or percentage of correct predictions of the classifier. For k -folder cross-validation, k confusion matrices are generated and summed to construct the overall confusion matrix.

3.4 Global Sensitivity Analysis

Among various sensitivity analysis methods, Global Sensitivity Analysis (GSA) has the ability to collect information from the entire input domain instead of local points, which makes the quantitative measurement of the sensitivity over the entire domain possible. The primary idea of GSA is to divide the uncertainty or variance of model output into different sources of uncertainty or variance of model inputs, and the contributions from different sources are

341 quantified by sensitivity indices.

342 The most sensitive parameter is identified through the GSA framework in both sense of
 343 self-contribution and parameter interactions, and insensitive parameters are set as constant in
 344 the view of computational complexity. In the view of multiple potential failure mechanisms,
 345 GSA provide the ability to understand the dominant mechanisms behind the complex input-
 346 output relationships over the input space, and eliminate the trivial influence from insensitive
 347 parameters. The parameter sensitivities obtained from GSA are parameter space dependent,
 348 considering that failure mechanisms are dominant at different parameter subspaces, and GSA
 349 focuses on the output uncertainty over the desired parameter space.

350 3.4.1 Decomposition of Variance

351 Arbitrary target function $y = f(\mathbf{u}) = f(u^1, \dots, u^d)$ is decomposed as [49]

$$f(u^1, \dots, u^d) = f_0 + \left[\sum_{i=1}^d \phi_i(u^i) + \sum_{i_1=1}^{d-1} \sum_{i_2=i_1+1}^d \phi_{i_1, i_2}(u^{i_1}, u^{i_2}) + \dots + \phi_{1, 2, \dots, d}(u^1, \dots, u^d) \right] \quad (32)$$

where

$$\left\{ \begin{array}{l} f_0 = \int f(\mathbf{u}) \prod_{i=1}^d [p_i(u^i) du^i] = E(y) \quad (33) \\ \phi_i(u^i) = \int f(\mathbf{u}) \prod_{j \neq i} [p_j(u^j) du^j] - f_0 = E_{u^{-i}}(y|u^i) - f_0 \quad (34) \\ \phi_{i_1, i_2}(u^{i_1}, u^{i_2}) = \int f(\mathbf{u}) \prod_{j \neq i_1, i_2} [p_j(u^j) du^j] - \phi_{i_1}(u^{i_1}) - \phi_{i_2}(u^{i_2}) - f_0 \quad (35) \end{array} \right.$$

352 with $p_i(u^i)$ the probably distribution function (PDF) of u^i , thus $\int f(\mathbf{u}) \prod_{j \neq i} [p_j(u^j) du^j]$ is
 353 $E_{u^{-i}}(y|u^i)$ by definition. The constant f_0 is the expectation value of y . $\phi_i(u^i)$ represents the
 354 contribution of u_i to f only from itself, while $\phi_{i_1, i_2}(u^{i_1}, u^{i_2})$ represents the contribution from
 355 the interaction between u^{i_1} and u^{i_2} to f . The variance of y is obtained:

$$\text{Var}(y) = \int [f(u^1, \dots, u^d) - f_0]^2 \prod_{i=1}^d [p_i(u^i) du^i] \quad (36)$$

356 The parameters are taken to be independent (i.e., uncorrelated) random variables, which
 357 ensures the uniqueness of the the decompositions stated above [49].

358 3.4.2 Sensitivity Index

359 Sensitivity index is a measurement of the parameter sensitivity/importance to the model out-
 360 put variance. Sensitivity indices quantify the contribution of the parameter itself and the
 361 interaction between the parameter and other parameters across the entire input domain. The

362 first order index quantifies the contribution to output variance from the parameter itself, i.e.,
 363 the first order terms in Eq. 32, while total effect index also includes the higher order terms in
 364 Eq. 32, i.e., the interactions between the parameter and the other parameters. Substituting
 365 Eq. 36 into Eq. 32:

$$1 = \sum_{i=1}^d S_i + \sum_{i_1=1}^{d-1} \sum_{i_2=i_1+1}^d S_{i_1 i_2} + \dots + S_{1,2,\dots,d} \quad (37)$$

366 where the first order sensitivity index S_i is defined [49]:

$$S_i = \frac{\text{Var}(\phi_i(u^i))}{\text{Var}(y)} = \frac{\text{Var}_{u^i}(E_{u^{-i}}(y|u^i))}{\text{Var}(y)} \quad (38)$$

367 The notation u^{-i} represents all possible u^j for $j \neq i$. $\text{Var}_{u^i}(E_{u^{-i}}(y|u^i))$ is the variance of the
 368 expectation of y given u^i . $E_{u^{-i}}(y|u^i)$ is computed by varying u^{-i} for a fixed u^i , whereas
 369 $\text{Var}_{u^i}(E_{u^{-i}}(y|u^i))$ is calculated by varying u^i .

370 The total effect index, S_i^T , is expressed as:

$$S_i^T = 1 - \frac{\text{Var}_{u^{-i}}(E_{u^i}(y|u^{-i}))}{\text{Var}(y)} \quad (39)$$

371 where $E_{u^i}(y|u^{-i})$ is the expectation of y given u^{-i} . The total effect index S_i^T is the sum of
 372 first order index and corresponding higher order terms [18]:

$$S_i^T = S_i + \sum_j S_{ij} + \sum_{j_1=1}^{d-1} \sum_{j_2=j_1+1}^d S_{i j_1 j_2} + \dots + S_{1,2,\dots,d} \quad (40)$$

373 where S_{ij} represents the interaction between u^i and u^j , $S_{i j_1 j_2}$ represents the interactions be-
 374 tween u^i , u^{j_1} and u^{j_2} , $S_{1,2,\dots,d}$ is the interactions between all input parameters. The interaction
 375 terms in the total effect index is non-zero despite the fact that the parameter distributions are
 376 taken to be uncorrelated [28]. The numerical computation of sensitivity indices is performed
 377 using the Monte-Carlo based procedure proposed by Saltelli et al. [45]. The sum of total effect
 378 indices have the following property:

$$\sum_{i=1}^d S_i^T \geq 1 \quad (41)$$

379 This is due to the fact that the interaction between u^i and u^j is accounted for in both S_i^T and
 380 S_j^T .

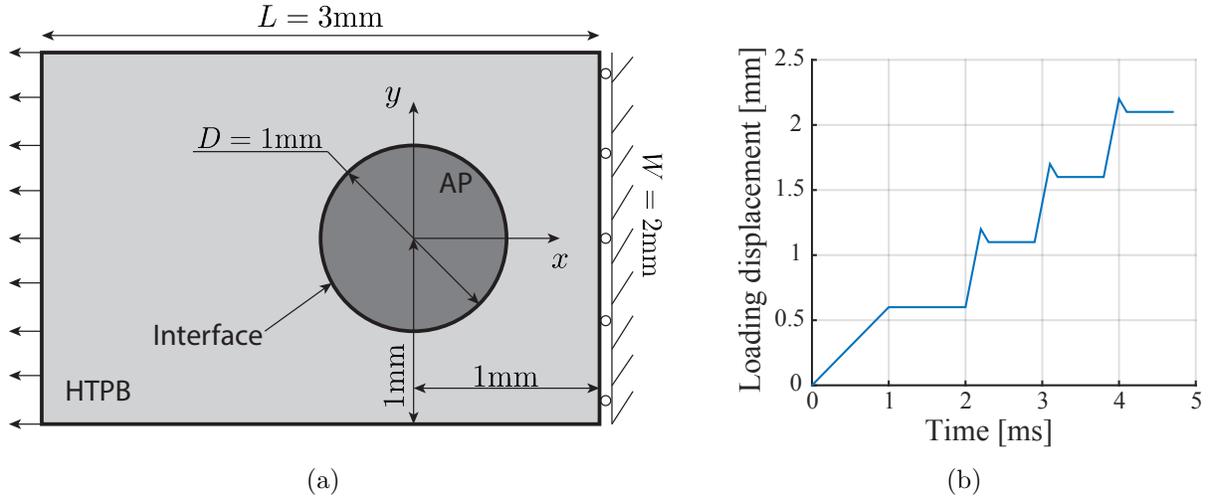


Figure 7: Dynamic response of a single AP particle reinforced HTPB. (a) The geometry and boundary conditions; (b) the loading history.

4 Numerical Examples

The sensitivity analysis framework has been exercised to study two cases: (1) The dynamic response of a single AP particle embedded in HTPB binder. Particularly, the investigation focuses on the sensitivity of interfacial separation observed under impact loading to the parameters that describe the constitutive behavior of the binder and the interface. (2) The dynamic response of a multi-particle mesostructure subjected to ultrasonic vibration loading to study the size and morphology sensitivity of the temperature rise within the mesostructure.

4.1 Material Property Sensitivity

The dynamic response of energetic materials is sensitive to the particle/binder interface, since the interfacial separation, upon debonding, leads to temperature rise induced by frictional heating and dissipation within the binder. In the current numerical study, material properties, especially interfacial parameters, are investigated using the proposed framework.

4.1.1 Mesoscale Model

In this study, a fixed mesostructure that consists of a single AP particle embedded in the HTPB binder is considered. The mesoscale geometry and corresponding boundary conditions are illustrated in Fig. 7a. The 3 mm \times 2 mm rectangular composite specimen includes a circular AP particle with a diameter of 1 mm. One edge of the specimen is fixed, while high rate impact loading is applied along the other edge. The displacement-controlled loading profile is shown in Fig. 7b, which consists of four step loadings. The loading rate in the first step is 600 mm/s, whereas the remaining three are applied at 3000 mm/s rate. The loading profile

Table 2: Binder and particle fixed parameters.

Parameter	ρ_h	ν_h	C_{V_h}	T_{ref}	A	B	E_{AP}	ρ_{AP}
Unit	g/mm^3		$\text{mJ}/\text{mm}^3\text{K}$	K			MPa	g/mm^3
Value	$0.95 \cdot 10^{-3}$	0.45	1.987	253	-15	102	$1.95 \cdot 10^4$	$1.95 \cdot 10^{-3}$
Parameter	$C_{V_{AP}}$	ν_{AP}	p_3	p_4	q_1	q_2	q_3	q_4
Unit	$\text{mJ}/\text{mm}^3\text{K}$		MPa	MPa	ms	ms	ms	ms
Value	2.121	0.3	25	13	$2 \cdot 10^{-5}$	$2 \cdot 10^{-4}$	$2 \cdot 10^{-3}$	$2 \cdot 10^{-2}$

is similar to a tensile Kolsky bar test that includes a quasi-static preload, followed by the applied load amplitude and multiple reflections. Under the applied loading, the AP particle is assumed to behave elastically. The HTPB binder undergoes viscoelastic deformation, and the interface progressively debonds based on the cohesive zone model described above. The response function of interest is the time to onset of interface separation defined as the time when the maximum separation reaches half of the particle diameter.

The AP particle Young’s modulus and Poisson’s ratio are obtained from [29]. The density ρ_{AP} is 1.95×10^{-3} , and the specific heat capacity per unit volume $C_{V_{AP}}$ is $2.121 \text{mJ}/\text{mm}^3\text{K}$ [16].

Fifteen material parameters fully describe the thermo-mechanical deformation in the binder. The viscoelastic response is approximated using a four-component Prony Series model. The parameter set is split into two subsets. The first set consists of those parameters a-priori considered to be insensitive with respect to the response function, or are fixed. The density of the HTPB binder, ρ_h , is $0.95 \times 10^{-3} \text{g}/\text{mm}^3$. The Poisson’s ratio of HTPB is taken to be 0.45, the same as that of Sylgard 184 [46, 53]. The specific heat capacity per unit volume, C_{V_h} is $1.987 \text{mJ}/\text{mm}^3\text{K}$ [16]. The reference temperature T_{ref} and temperature shift factors A , B are kept consistent with Ref. [20]. The values of the fixed parameters are summarized in Table 2. The remainder of the parameters (i.e., the second set) are taken to be sensitive to the response function and employed as variables in the sensitivity study. The specific distributions of the parameters are unknown (due to lack of sufficient experiments to characterize the distributions). The parameters are therefore assumed to follow uniform distributions, which only require lower and upper bounds. The parameter ranges within the second set are identified based on experimental data and calibrated from stress-strain curves at different temperature and loading rates. Figure 8 shows the identification of the upper and lower bounds for the viscoelastic parameters (p_1 , p_2 , G_{h_∞}) by fitting bounds to the HTPB experimental stress-strain curves based on the data from Ref. [4]. The parameter ranges for the viscoelastic parameters employed in the sensitivity analysis are shown in Table 3.

Ten parameters describe the thermo-mechanical behavior and the progressive debonding along the particle-binder interface. Similar to the binder case, the parameter set is split into two subsets. The first set consists of those parameters a-priori considered to be insensitive

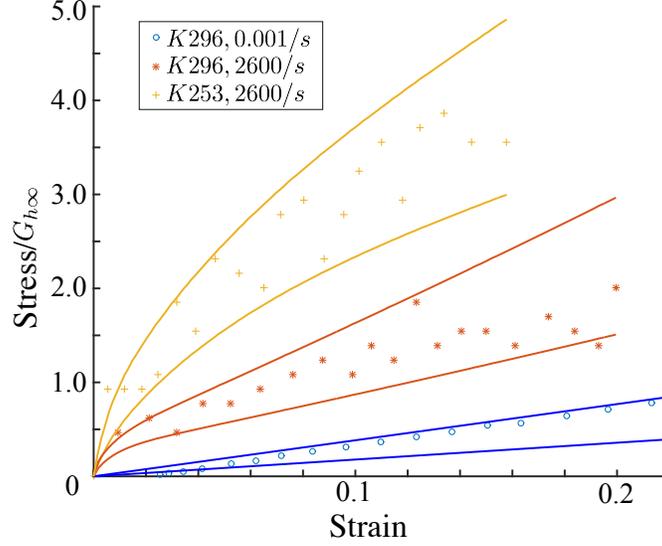


Figure 8: Prony series ranges calibration with experimental data.

Table 3: Parameter boundaries in material property sensitivity study.

Parameter	p_1	p_2	$G_{h\infty}$	T_{\max}	Δ_{cn}	Δ_{ct}	μ_0	w_{int}	ρ_{int}
Unit	MPa	MPa	MPa	MPa	mm	mm		mm	g/m^3
Upper bd	80	48	$1.95 \cdot 10^{-3}$	2.9	5	0.25	0.3	0.3	$3 \cdot 10^{-3}$
Lower bd	65	27	$0.95 \cdot 10^{-3}$	1.35	0.5	0.1	0.2	0.2	$2 \cdot 10^{-4}$

Table 4: Interfacial fixed parameters

Parameter	η_0	ϵ	β_s	$C_{V_{\text{int}}}$
Unit			MPa/mm	$\text{mJ}/\text{mm}^3\text{K}$
Value	0.014	10^{-5}	1900	2.054

430 with respect to the response function, or are fixed. The elastic limit η_0 is reported in [20]. The
431 regularization parameter ϵ , which is introduced from the numerical efficiency point of view,
432 is determined as 1×10^{-5} by a numerical parametric study in which the error caused by the
433 regularization is not greater than 0.2% compared to the classical Coulomb law. The interfacial
434 stiffness β_s is experimentally obtained by measuring the slope of the stress-strain curve [20].
435 The interfacial heat capacity $C_{V_{\text{int}}}$ is taken as the average of the two material phases (AP and
436 HTPB), i.e., $C_{V_{\text{int}}} = 2.054 \text{mJ}/\text{mm}^3\text{K}$. The values of the fixed parameters are summarized in
437 Table 4. The remainder of the parameters are taken to be sensitive to the response function
438 and employed as variables in the sensitivity study. The maximum traction T_{\max} , the critical
439 normal separation Δ_{cn} , the critical tangential separation Δ_{ct} , the interfacial width w_{int} , the
440 interfacial density ρ_{int} , and the friction coefficient μ_0 are included as variables in the sensitivity
441 study. Parameter ranges of the cohesive zone model are selected based on the properties of
442 the PBXs investigated in Refs. [1, 15, 20, 36, 51, 63] and listed in Table 3.

Table 5: Overall accuracies of classifiers

Classifier	Linear SVM	Simple Tree	Coarse KNN ¹	LD ²
Accuracy	94.4%	81.4%	95.3%	90.4%
Prediction Speed [obj/s ⁴]	~ 28000	~ 96000	~ 33000	~ 72000
Classifier	Quadratic SVM	Medium Tree	Medium KNN	QD ³
Accuracy	98.2%	96.6%	96.6%	91.0%
Prediction Speed [obj/s]	~ 25000	~ 120000	~ 57000	~ 52000

¹ k -Nearest Neighbor

²LD represents linear discriminant

³QD represents quadratic discriminant

⁴obj/s represents objective (or number of function evaluations) per second

In summary, three binder parameters (i.e., p_1 , p_2 and G_{h_∞}) and six interfacial parameters (i.e., T_{\max} , Δ_{cn} , Δ_{ct} , w_{int} , ρ_{int} , and μ_0) for a total of nine parameters are included in the current sensitivity study.

4.1.2 Local GP Models and Classifier

Under the applied step-wise loading, interface separation (as a function of material parameters) tends to cluster near the load steps. This results in a multi-modal distribution of time-to-separation as further discussed below. In order to accurately approximate this behavior using surrogate modeling, we employed the piecewise continuous GP model (Eq. 27). The classifier SVM provides higher accuracy than some of the other classifiers (e.g. decision tree, k -Nearest Neighbor Classifier, discriminant analysis) at the cost of added computational complexity, as shown in Table 5.

A five-part GP model has been constructed to predict the sensitivity of the nine material parameters on the failure behavior of the composite. The five-part model corresponds to the parameter subspaces that result in failure near the four step loads, in addition to the subspace that result in no interface separation.

In Fig. 9, Von Mises stress contours are plotted from four representative simulations at the state of interface failure (i.e., separation time). The contour plots indicates a change in interface separation characteristics in addition to separation time as a function of constituent parameters. The proposed five-part GP model is trained by a suite of training points that are obtained through the dynamic analysis of the mesostructure using the finite element method with randomly sampled parameters using the stratified sampling method. The accuracy of the GP model has been assessed by verifying the convergence as a function of the number of training points, and the number of prediction points, as well as by verifying the accuracy of subspace classification through the confusion matrix.

Figure 10a illustrates the convergence of the GP model as a function of the number of

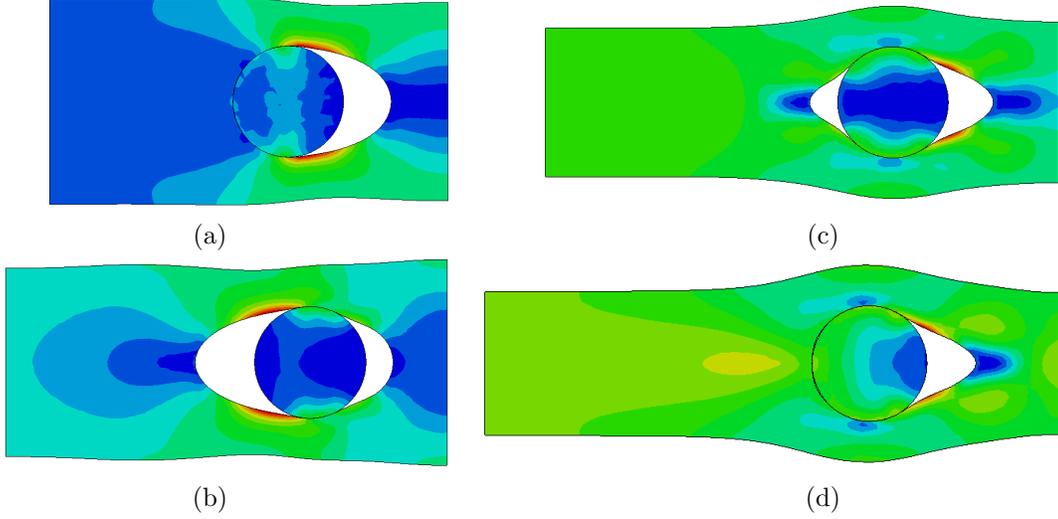


Figure 9: Von Mises stress contours of single particle system failure at (a) $t = 1.04$ ms; (b) $t = 2.12$ ms; (c) $t = 3.08$ ms; (d) $t = 3.932$ ms.

468 training points employed. The figure shows the prediction of time-to-separation PDFs based
 469 on three piecewise continuous GP models trained by 256, 512, 5120 mesoscale simulations.
 470 The PDFs were generated using 20 million prediction points for all three models. While the
 471 model trained by 256 training points demonstrates significant discrepancy, the models trained
 472 by 512 and 5120 indicates close match. In the remainder of the numerical study, we employ
 473 the GP model trained by 5120 training points. Figure 10b shows the convergence of the GP
 474 model predictions as a function of the number of prediction points. The figure demonstrates
 475 that the GP model predictions converge after a relatively large number of prediction points are
 476 employed (~ 2 million). It is important to note that the figure illustrates those prediction points
 477 that lead to separation. The distributions with low number of prediction points demonstrates
 478 a higher ratio of cases in which separation does not initiate.

479 The performance of the SVM classifier is assessed by the confusion matrix which is gener-
 480 ated by a 10-folder cross-validation as shown in Fig. 11.

481 In Fig. 11b, all the diagonal terms (indicating correct classification) are greater than 90%,
 482 which indicates that the classifier correctly predicts the appropriate parameter subspace with
 483 over 90% accuracy. Increasing accuracy is observed as the number of training points increases
 484 from 512 (Fig. 11a) to 5120 (Fig. 11b) employing quadratic SVM classifier.

485 4.1.3 Global Sensitivity Analysis

486 The parameter boundaries which define the parameter space of interest are provided in Table 3.
 487 The most sensitive parameters using the first order sensitivity and total effect sensitivity are
 488 identified according to the indices of each parameter.

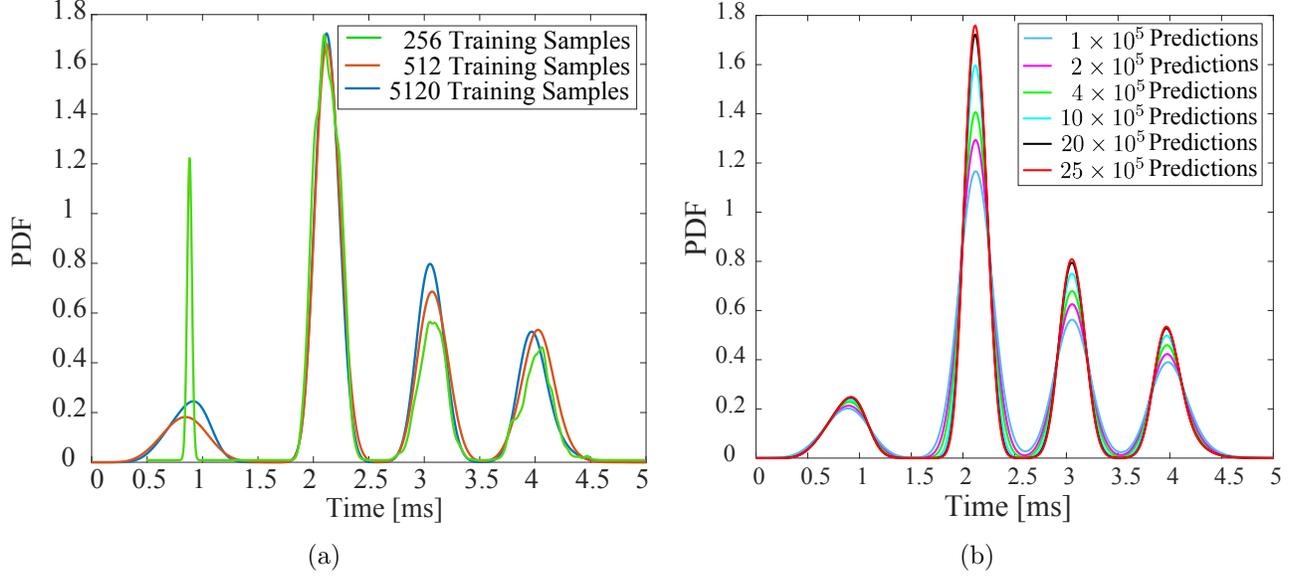


Figure 10: Convergence study of GP model with respect to (a) the number of training samples with prediction distributions generated with 2 million samples; and (b) the number of prediction points with the surrogate trained with 5120 samples.



Figure 11: Confusion matrix. Class i : separation initiation near the i th step load ($i \leq 4$). Class 5: no separation observed. (a) confusion matrix trained with 512 samples. (b) confusion matrix trained with 5120 samples.

489 Sensitivity indices, the quantitative representation of parameter contribution to interfacial
 490 separation, are shown in Fig. 12a. Interface strength T_{\max} is the dominant parameter based on
 491 self-contribution as well as parameter interactions. The critical normal separation and G_{h_∞} are
 492 also identified as significant uncertainty sources in this system. The higher order interactions
 493 between multiple factors significantly increase the sensitivities of Δ_{cn} and G_{h_∞} . Compared to
 494 critical normal separation, the critical tangent separation is an insensitive parameter, despite
 495 the presence of shear stress concentration as reported in [20]. Prony Series and other interfa-
 496 cial parameters are observed to be insensitive factors as well. Figure 12b demonstrates the
 497 convergence of the sensitivity measures as a function of the number of prediction samples. The
 498 figure indicates convergence with approximately 20 million prediction samples.

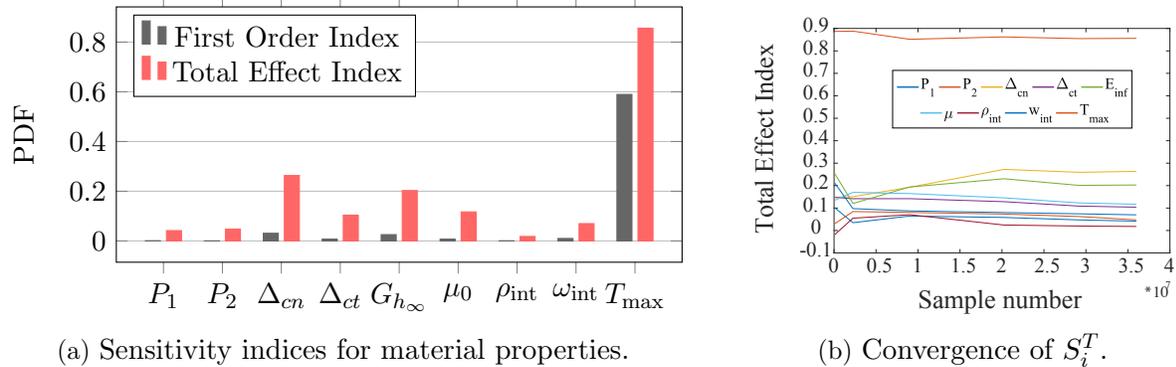


Figure 12: Convergence study of GP model with respect to (a) the number of training samples with prediction distributions generated with 20 million samples; and (b) the number of prediction points with the surrogate trained with 5120 samples.

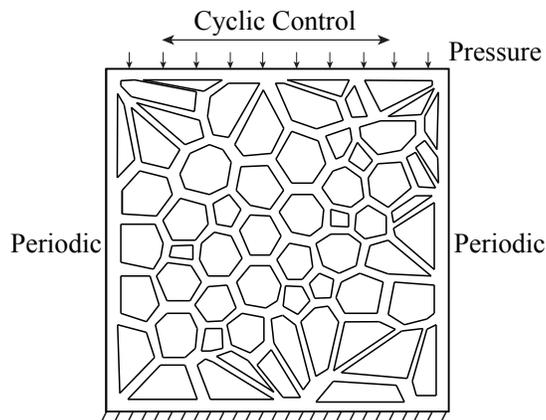


Figure 13: Geometry and loading conditions.

4.2 Mesoscale Geometry Sensitivity

The local temperature rise (“hot spots”) induced by local mesostructure response in multiple particle systems under dynamic loading is of interest, since hot spots are among the failure mechanisms in energetic materials. In the current numerical study, the mesoscale geometry characteristics of the particulate composite, for example, particle size and particle shape, are parameterized and investigated using the GSA framework.

4.2.1 Mesoscale Model

In this numerical study, the thermo-mechanical dynamic behavior of a multiple particle AP-HTPB composite specimen under ultrasonic vibration loading is investigated to assess the contribution of geometric parameters on the temperature rise in the specimen. The specimen is idealized as a $3 \text{ mm} \times 3 \text{ mm}$ rectangle with multiple polygons (idealized particles) embedded as shown in Fig. 13. The bottom edge is fixed, and the periodic boundary condition is applied

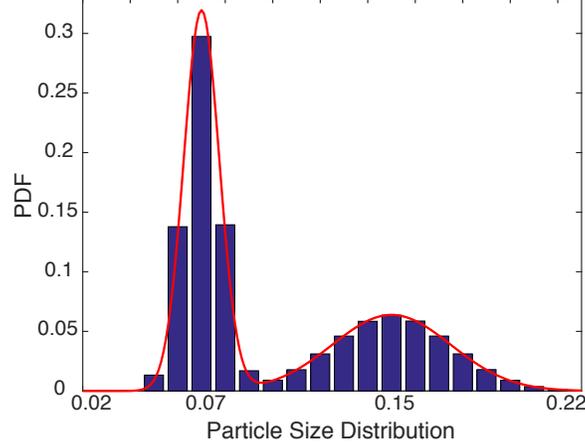


Figure 14: Illustration of particle size distribution. Above PDF is subjected to a bimodal distribution with $\mu_1 = 0.15 \text{ mm}^2$, $\mu_2 = 0.07 \text{ mm}^2$, $\sigma_1 = 0.025 \text{ mm}^2$, $\sigma_2 = 0.0075 \text{ mm}^2$, $w_1 = 0.6$.

511 on the side edges. Static pressure, $p = 200 \text{ KPa}$, and a harmonic loading, $d_h = A_v \cdot \sin(2\pi f \cdot t)$,
 512 are both applied on the top surface as external excitation. $A_v = 0.03 \text{ mm}$ is the loading
 513 amplitude, and $f = 40 \text{ kHz}$ is the ultrasonic frequency.

514 Mesoscale geometry of particulate composites is defined by the particle size distribution,
 515 the particle shape distribution, and the particle area fraction. The mesoscale geometry of AP-
 516 HTPB system is assumed to be controlled by the above parameterized distribution parameters
 517 and constructed by Neper [37], a polycrystalline microstructure generation software. The
 518 geometry of the particles obtained from Neper is adjusted to reach the desired area fraction,
 519 and the binder model is applied on the gap between the AP particles to include the interaction
 520 of particles through the soft HTPB binder. The cohesive zone element is inserted between
 521 particle and binder to reproduce the interfacial behavior.

522 In the context of numerical verifications, a bimodal particle size distribution is assumed [2]:

$$f(A) = w_1 \cdot N(\mu_1, \sigma_1) + (1 - w_1) \cdot N(\mu_2, \sigma_2) \quad (42)$$

523 where w_1 is the weight for the first mode. μ_1 and σ_1 are the mean and variance of the normal
 524 distribution of the first mode. μ_2 and σ_2 are the mean and variance of the second mode, as
 525 illustrated in Fig. 14.

526 Sensitivities of five mesostructure parameters that define the mesoscale geometry (particle
 527 size distribution means μ_1 , μ_2 , bimodal weight w_1 , particle sphericity distribution mean μ_s ,
 528 and area fraction η) are investigated. Similar to the previous example, uniform distribution
 529 is assumed for each parameter. The values of parameter ranges are selected according to
 530 experimental and numerical energetic mesostructure data available in the literature [1, 56,

Table 6: Parameters boundary in mesoscale geometry sensitivity study.

Parameter	μ_1	μ_2	μ_s	ω_1	η
Unit	mm ²	mm ²			
Upper bd	0.2	0.07	0.9	1.0	0.9
Lower bd	0.1	0.04	0.8	0.5	0.7

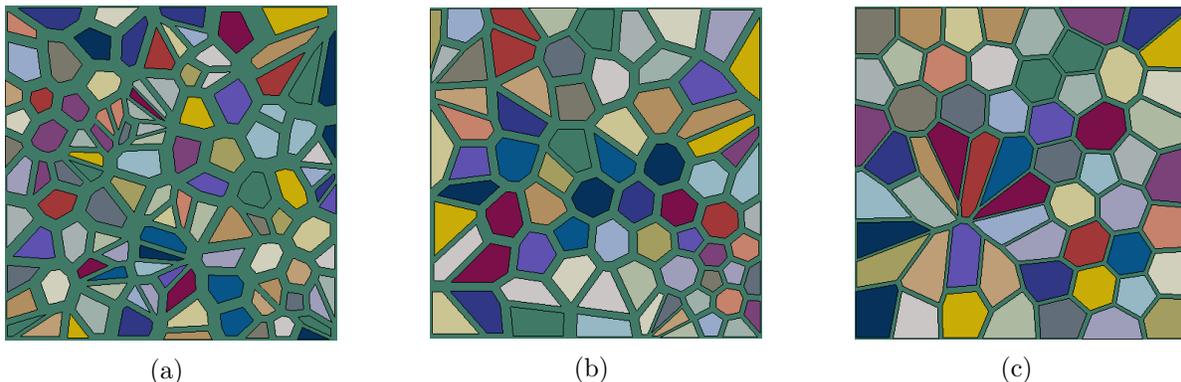


Figure 15: Examples of mesoscale geometry with size and shape parameters at (a) lower boundaries; (b) mean values; (c) upper boundaries in Table 6.

Table 7: Fixed parameters in mesoscale geometry sensitivity study.

Parameter	T_{\max}	Δ_{cn}	Δ_{ct}	μ_0	w_{int}	ρ_{int}	$G_{h\infty}$
Unit	MPa	mm	mm		mm	g/m ³	MPa
Value	2.91	0.11	0.264	0.2	10 ⁻³	1.45 · 10 ⁻³	2.5

27, 17, 9, 30, 10, 61, 21] and listed in Table 6. Variations of these five parameters directly influence the mesoscale geometry, and three examples in Fig. 15 are employed to demonstrate the influence of the parameters on the resulting mesoscale geometry. The variance of the first mode and second mode, σ_1 and σ_2 are kept as 0.025 mm and 0.0075, respectively. The variance of sphericity σ_s is selected as 0.15.

Material parameters except the Prony Series (i.e., AP particle properties, interface parameters) set in the previous example are used in the current study as well. Six interfacial parameters and the steady-state shear modulus employed before are assumed to be constants and shown in Table 7. Six pairs of Prony Series are employed to represent the viscoelastic behavior of HTPB binder and calibrated from experimental data [4]. The calibrated stress-strain curves using six pairs of Prony Series are plotted in Fig. 16, and corresponding moduli and relaxation times are shown in Table 8.

In order to ensure that the element discretization is sufficient to capture the dynamic response, a mesh convergence study with respect to the target response functions (i.e., temperature) is performed. For a representative case as shown in Fig. 17a, the specimen is deformed

Table 8: Calibrated six pairs of prony series.

Parameter	p_1	p_2	p_3	p_4	p_5	p_6
Value	33	30	25	13	8	3
Parameter	q_1	q_2	q_3	q_4	q_5	q_6
Value	$1.04 \cdot 10^{-7}$	$2.1 \cdot 10^{-5}$	$1.66 \cdot 10^{-3}$	$1.05 \cdot 10^{-2}$	$5 \cdot 10^{-2}$	$2.1 \cdot 10^{-1}$

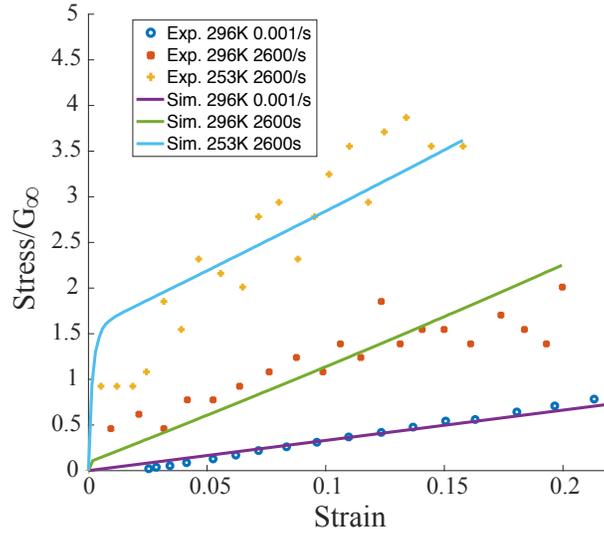
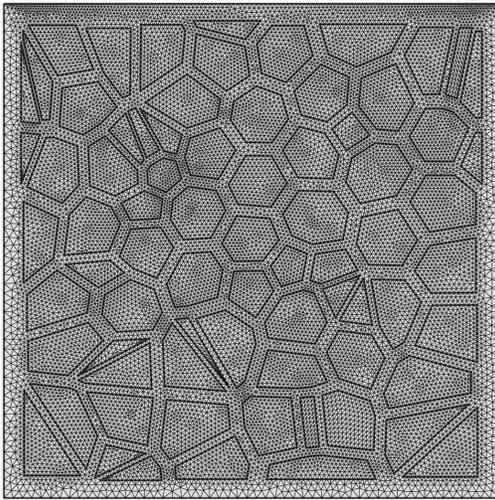
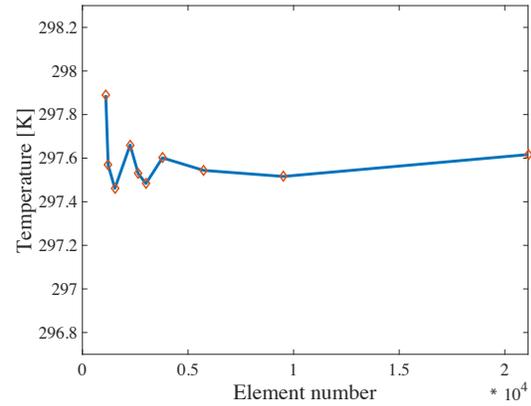


Figure 16: Prony Series Calibration



(a)



(b)

Figure 17: Mesh convergence study of the representative case. (a) Mesh of the representative mesostructure with 21104 elements; (b) Element number convergence.

546 under ultrasonic loading for 1 ms. The maximum temperature after 1 ms as the response func-
 547 tion of the mesh convergence study changes with mesh refinement and its convergence trend
 548 is plotted in Fig. 17b. When the number of elements reaches to around 6,000, the maximum

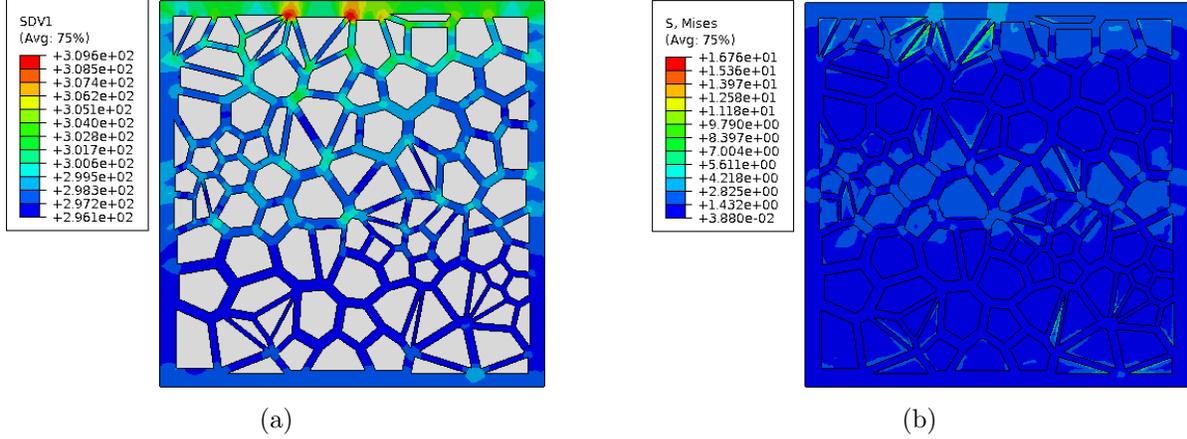


Figure 18: Example of mesoscale FE simulations of multiple particle systems. (a) Temperature contour of binder at $t = 10$ ms; (b) Von Mises stress contour at $t = 10$ ms.

549 temperature in the specimen after the ultrasonic excitation is loaded for 1 ms is stable. All
 550 forward mesoscale simulations employ a level of discretization similar to the converged model
 551 described in this study.

552 4.2.2 GP Surrogate Model

553 Although high temperature may exist at multiple local sites within the mesostructure, as shown
 554 in Fig. 18, the maximum temperature within the specimen is chosen as the response function.
 555 A GP surrogate is trained to substitute the complex mesoscale FE simulation (Fig. 18). The
 556 convergence with respect to the number of training samples is guaranteed by constructing four
 557 GP models with 32, 64, 96, and 128 mesoscale simulations as training points. Four prediction
 558 distributions constructed with GP models trained by different sample sizes are compared and
 559 the convergence is observed, as shown in Fig. 19a. The PDFs were generated using forty
 560 million prediction points for all four models. These four distributions are not identical, but
 561 three of them (produced by GP models trained with 64, 96, 128 samples) are similar and can
 562 be regarded as converging. In current study, the GP model with 64 training simulations is
 563 employed.

564 Similar to the previous study, the convergence of prediction distributions should be guar-
 565 anteed before checking the convergence of the number of training samples. Forty million
 566 prediction points are necessary to generate the converged prediction distribution, as shown in
 567 Fig. 19b.

568 4.2.3 Global Sensitivity Analysis

569 Five mesoscale geometry parameters that represent the mesostructure characteristics are in-
 570 vestigated through the proposed sensitivity analysis framework. Sensitivity indices quantifying

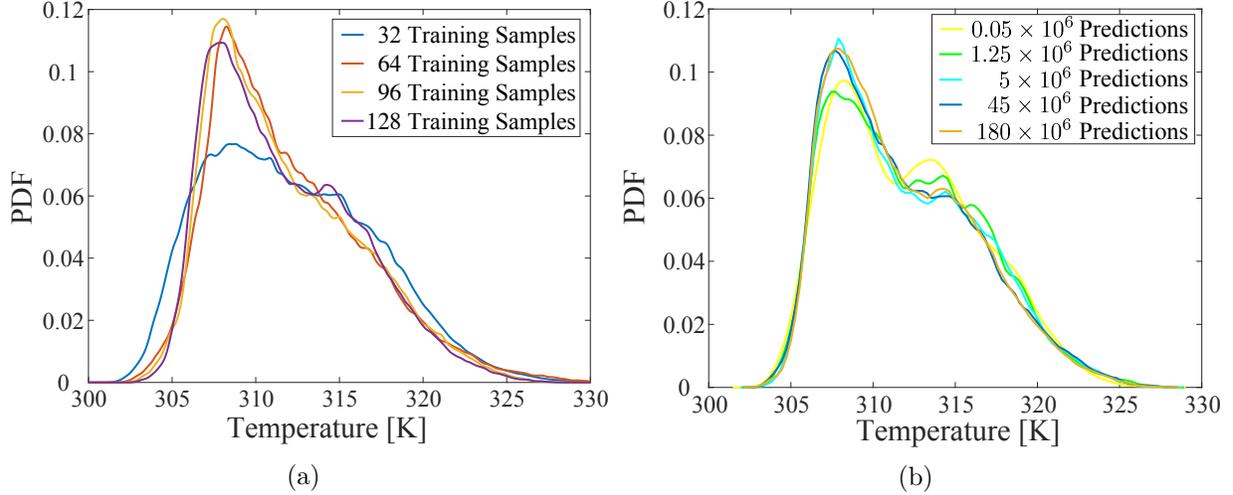
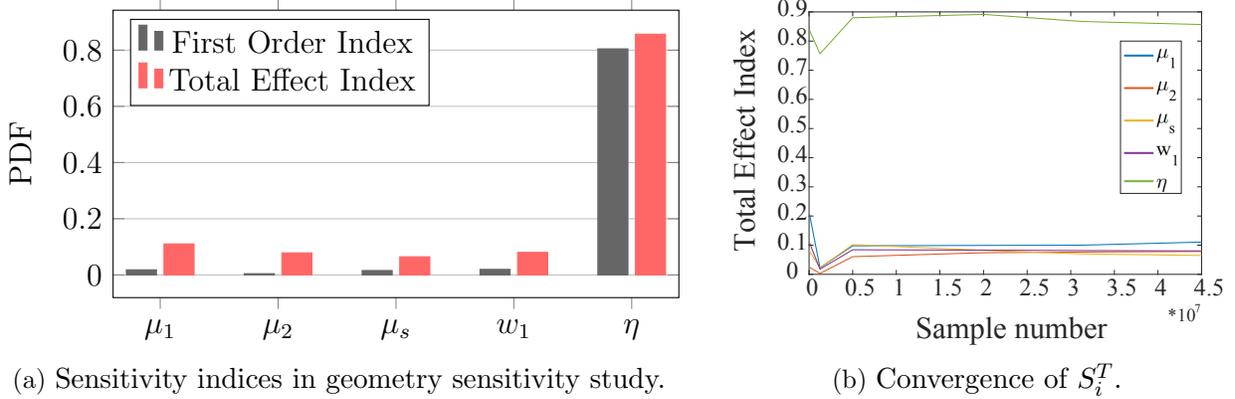


Figure 19: Convergence study of GP model with respect to (a) the number of training samples with prediction distributions generated with 40 million samples; (b) the number of prediction points with the surrogate trained with 128 samples.



(a) Sensitivity indices in geometry sensitivity study.

(b) Convergence of S_i^T .

Figure 20: Convergence study of GP model with respect to (a) the number of training samples with prediction distributions generated with 45 million samples; and (b) the number of prediction points with the surrogate trained with 5120 samples.

571 contributions from parameters and interactions are obtained, and the dominant parameter is
 572 identified.

573 The sensitivity indices obtained from the GSA framework are shown in the Fig. 20a. The
 574 results indicate that the area fraction is the dominant parameter that contributes to the tem-
 575 perature rise in the dynamic simulations. The dominant effect of area fraction is primarily
 576 due to the higher stress concentrations observed in particles in close proximity to one another.
 577 It is interesting to note that the sphericity parameter appears to have little influence on the
 578 localized heating characteristics. This result is surprising since higher stresses are expected
 579 near low sphericity particles. This result is attributed to the fact that the parameter range for
 580 sphericity is kept relatively narrow in the sensitivity study. This is to avoid the need to employ

581 very fine discretization near narrow edges of the particles. From the physical perspective, it
582 can be speculated that very fine asperities of the particles break off at the early stages of
583 the dynamic loading, rounding the particle geometries prior to the onset of localized heating.
584 Comparing the total and first order sensitivity indices, the contributions of the parameter
585 interactions appear to be significant for particle mean sizes, the weight and sphericity expect-
586 ation parameters. The larger particle mean size exhibits larger influence than the smaller
587 particle mean size to the maximum temperature within the particulate composite. Figure
588 20b demonstrates the convergence of the sensitivity measures as a function of the number of
589 prediction samples. The figure indicates convergence with approximately 20 million prediction
590 samples.

591 **5 Conclusion**

592 This manuscript presented a new global sensitivity analysis framework for problems that ex-
593 hibit discontinuous response functions. The proposed approach has been applied to investigate
594 the role of material and morphological properties of particulate energetic composite materials
595 subjected to dynamic loading conditions. In particular, the following conclusions are drawn:
596 (1) the piece-wise continuous GP model with SVM classifier to characterize the continuous sub-
597 domains provide a computationally accurate and efficient approach to characterize sensitivities
598 in the presence of response function discontinuities; (2) Interface strength and particle volume
599 fraction are found to be the most influential parameters in the loading regime and mesoscale
600 morphologies investigated in this study.

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