

1 Variational multiscale enrichment method for dynamic response of
2 hyperelastic materials at finite deformation

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8 **Abstract**

9 In this manuscript, we extend the variational multiscale enrichment (VME) method to
10 model the dynamic response of hyperelastic materials undergoing large deformations. This
11 approach enables the simulation of wave propagation under scale-inseparable conditions, in-
12 cluding short-wavelength regimes, while accounting for material and geometric nonlinearities
13 that lead to wave steepening or flattening. By employing an additive decomposition of the
14 displacement field, we derive multiscale governing equations for the coarse- and fine-scale prob-
15 lems, which naturally incorporate micro-inertial effects. The framework allows the discretiza-
16 tion of each unit cell with a patch of coarse-scale elements, which is essential to accurately
17 capture wave propagation in short-wavelength regimes. An operator-split procedure is used to
18 iteratively solve the semi-discrete equations at both scales until convergence is achieved. The
19 coarse-scale problem is integrated explicitly, while the fine-scale problem is solved using either
20 explicit or implicit time integration schemes, including both dissipative and non-dissipative
21 methods. Numerical examples demonstrate that multiscale dissipative schemes effectively sup-
22 press spurious oscillations. The multiscale framework was applied to investigate how material
and geometric nonlinearities, along with elastic stiffness contrast in heterogeneous microstruc-
tures, influence key wave characteristics such as dispersion, attenuation, and steepening. This
multiscale computational framework provides a foundation for studying the dynamic response
of architected materials.

23 *Keywords:* Multiscale modeling; Transient dynamics; Wave propagation; Hyperelasticity

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1 Introduction

Architected materials have attracted widespread interest in the research community due to their ability to achieve exceptional properties by tailoring geometric features across multiple length scales, often outperforming conventional materials in terms of mechanical, thermal, or functional performance. Energy-absorbing architected materials such as honeycombs and auxetic metamaterials demonstrate favorable impact resistance and deformation behavior, making them excellent candidates for crash protection and vibration damping applications (see e.g., [22, 58, 53, 18, 10]). Computational simulation of the transient dynamic response of structures composed of architected unit cells using direct numerical simulations (DNS) is computationally expensive and cumbersome, especially in scenarios involving large structural domains or complex microstructural features. As a result, there is a critical need to develop multiscale modeling techniques that can efficiently simulate the dynamic response of such structures while preserving accuracy.

Various homogenization approaches have been proposed to model the dynamic response of heterogeneous materials based on the assumption that the microstructural size is much smaller than the wavelength, commonly referred to as the scale separation limit. In the computational homogenization approach [49, 9, 35, 50], the fundamental idea is to characterize the material response locally at each quadrature point of the finite element discretization of the macroscopic domain by the numerical evaluation of a representative volume element (RVE) or a periodic unit cell. Nested initial-boundary value problems at the macro- and microscales are evaluated, with scale bridging relations that satisfy energy consistency between scales, i.e., the Hill-Mandel condition. Higher-order asymptotic homogenization approaches were developed to capture the dynamic homogenized response at the short-wavelength regime [1, 17, 31, 26, 27, 28]. Some alternative multiscale methods that do not rely on the assumption of scale separation, including the elastodynamic homogenization methods based on Willis' theory [52, 39, 41, 40, 38], multiscale finite element method [11, 12], and the method of computational continua [16, 14] have been used for wave propagation in short-wavelength regime. The computational homogenization framework has also been extended to lattice metamaterials, including both truss- and beam-based architectures [23, 24, 34], though most developments to date have primarily focused on quasistatic loading conditions.

The variational multiscale method (VMM) [30] is an alternative multiscale strategy that does not assume scale separation. This method is based on the additive split of the cardinal response field into coarse and fine-scale components, resulting in separate but coupled multiscale problems. Numerical efficiency is achieved by evaluating the fine-scale problem analytically (when an analytical form that approximates fine-scale response is known; e.g., see [46]). Problems that involve complex micro-morphologies and nonlinearities may not have

analytical forms to represent the fine-scale response with sufficient accuracy. This prompted VMM variants that treat the fine-scale problem numerically, such as the numerical subgrid upscaling method [2], the stochastic variational multiscale method [3, 21], and the variational multiscale enrichment method [47, 48, 55, 56, 57]. Of particular relevance is the spectral variational multiscale enrichment [29], which was developed to model the transient dynamics and wave propagation of phononic crystals and acoustic metamaterials. This study focused on the linear material behavior, and the effects of geometric and material nonlinearities on wave propagation were not considered.

Obtaining accurate solutions for transient dynamics or wave propagation problems in the short-wavelength regime (using either direct numerical simulations or multiscale methods) is a challenge. Numerical errors due to spatial and time discretizations using finite element methods and time integration schemes lead to artificial period elongations and amplitude decays, which manifest themselves as numerical dispersion and dissipation errors [4, 44]. A plethora of approaches have been developed to address the dispersion and dissipation errors, including the use of higher-order spatial discretizations [25, 20], finite element interpolations enriched with wave packets for multiscale wave propagation problems [32], or the use of spectral elements in the context of multiscale description [29].

Higher-order elements can improve accuracy but are often too expensive, motivating the use of time integration schemes with lower-order elements that incorporate controlled numerical dissipation [4, 33, 19]. The Bathe implicit method [5] addresses this by filtering out unresolved high-frequency modes while accurately integrating the resolvable ones, thereby reducing dispersion errors. Extending this idea, Noh and Bathe [43] proposed an explicit scheme with high-frequency dissipation that preserves second-order accuracy and produces the desired behavior for period elongations and amplitude decays, small for small time steps and rapidly increasing for larger ones. In contrast, the Tchamwa–Wielgosz scheme [51], though only first-order accurate, performs better than many classical explicit schemes [42, 13, 54, 45], but is still outperformed by Noh and Bathe’s method.

In this study, we present a robust multiscale simulation framework based on the variational multiscale enrichment (VME) method for modeling the transient dynamic response of hyperelastic composite materials undergoing large deformations. The approach employs an additive decomposition of displacement, velocity, and acceleration fields into coarse- and fine-scale components, enabling a consistent derivation of the multiscale governing equations within a Lagrangian setting. To accurately resolve wave propagation in short-wavelength regimes, the formulation allows each unit cell to be discretized into multiple coarse-scale elements. The coupled coarse- and fine-scale problems are solved iteratively using an operator-split procedure until convergence is achieved.

The semi-discrete multiscale equations are integrated explicitly using either the non-dissipative

97 central difference method or the dissipative explicit scheme of Noh and Bathe [43]. Although
 98 the VME formulation is designed for scale-inseparable problems, differences in characteris-
 99 tic time scales between coarse and fine discretizations can arise. In such cases, particularly
 100 when accuracy requirements permit larger time steps than the stability limits of the fine-scale
 101 problem, a mixed integration strategy can be adopted – explicit integration for the coarse-
 102 scale problem and implicit integration of the fine-scale problem, for example, with the Bathe
 103 implicit method [5]. Numerical investigations show that spurious oscillations appear in both
 104 direct numerical and multiscale simulations when using the central difference method, but
 105 these are effectively suppressed by dissipative integration schemes. The proposed multiscale
 106 framework is shown to capture wave dispersion, attenuation, and wave steepening driven by
 107 microstructural heterogeneity, unit cell size, and geometric and material nonlinearities. Fi-
 108 nally, the computational performance of different time integration schemes is assessed for
 109 microstructures with varying elastic modulus contrasts.

110 The remainder of this manuscript is organized as follows: Section 2 presents the formulation
 111 of the multiscale method. Section 3 provides the computational approximation of the resulting
 112 coarse- and fine-scale partial differential equations (PDEs) from the VME method, along with
 113 the time integration schemes for the multiscale problem. Further details about the evaluation
 114 of non-standard element-level matrices and vectors in multiscale discretization, and estimation
 115 of stable time increment for the multiscale problem are discussed. Section 4 provides examples
 116 of wave propagation in 1-D problems with homogeneous and heterogeneous microstructures.
 117 Section 5 discusses concluding remarks and future research directions.

118 2 Variational Multiscale Enrichment Formulation

119 In this section, we introduce the formulation of the proposed VME approach for modeling the
 120 dynamic response of hyperelastic materials. Let us denote a heterogeneous body with $\Omega \in \mathbb{R}^{n_{sd}}$
 121 ($n_{sd}=1,2$ or 3), an open and bounded domain composed of repeated unit cells. The governing
 122 equations for linear and angular momentum balance in the Lagrangian setting are as follows:

$$\nabla_{\mathbf{X}} \cdot \mathbf{P}(\mathbf{X}, t) + \mathbf{B}(\mathbf{X}, t) = \rho_0 \ddot{\mathbf{u}}(\mathbf{X}, t); \quad \mathbf{X} \in \Omega, \quad t \in [0, T] \quad (1a)$$

$$\mathbf{P}(\mathbf{X}, t) \mathbf{F}^T(\mathbf{X}, t) = \mathbf{F}(\mathbf{X}, t) \mathbf{P}^T(\mathbf{X}, t); \quad \mathbf{X} \in \Omega, \quad t \in [0, T] \quad (1b)$$

123 where \mathbf{X} denotes the Cartesian coordinates in the reference (undeformed) configuration, t the
 124 time coordinate, \mathbf{P} the First Piola-Kirchhoff stress, \mathbf{F} the deformation gradient, \mathbf{B} the body
 125 force, ρ_0 the mass density in the reference configuration, and \mathbf{u} the displacement field. $\nabla_{\mathbf{X}} \cdot$
 126 denotes the divergence operator in the reference configuration, and $(\ddot{\cdot})$ denotes the second time
 127 derivative.

128 The Dirichlet and Neumann boundary conditions are specified respectively on $\Gamma^u \subseteq \partial\Omega$
 129 and $\Gamma^t \subseteq \partial\Omega$, such that $\Gamma^u \cup \Gamma^t = \partial\Omega$ and $\Gamma^u \cap \Gamma^t = \emptyset$:

$$\mathbf{u}(\mathbf{X}, t) = \tilde{\mathbf{u}}(\mathbf{X}, t); \quad \mathbf{X} \in \Gamma^u \quad (2a)$$

$$\mathbf{P}(\mathbf{X}, t) \cdot \mathbf{N} = \tilde{\mathbf{T}}(\mathbf{X}, t); \quad \mathbf{X} \in \Gamma^t \quad (2b)$$

130 where \mathbf{N} denotes the outward unit normal in the reference configuration, $\tilde{\mathbf{u}}$ and $\tilde{\mathbf{T}}$ are the
 131 prescribed displacement and traction vectors, respectively. The initial conditions are specified
 132 as follows:

$$\mathbf{u}(\mathbf{X}, 0) = \hat{\mathbf{u}}(\mathbf{X}); \quad \mathbf{X} \in \Omega \quad (3a)$$

$$\dot{\mathbf{u}}(\mathbf{X}, 0) = \hat{\mathbf{v}}(\mathbf{X}); \quad \mathbf{X} \in \Omega \quad (3b)$$

133 in which $\hat{\mathbf{u}}$ and $\hat{\mathbf{v}}$ denote the prescribed displacement and velocity fields respectively at $t = 0$.

134 At any given material point, \mathbf{X} , the constituent material is taken to follow a hyperelastic
 135 constitutive law. The first Piola-Kirchhoff stress tensor is given by:

$$\mathbf{P} = \frac{\partial \psi(\mathbf{F})}{\partial \mathbf{F}}, \quad (4)$$

136 where ψ denotes the strain energy density function. The angular momentum balance given by
 137 Eq. (1b) is satisfied by the objectivity of the strain energy density function.

138 The weak form of the linear momentum balance (Eq. (1a)) with the boundary conditions
 139 in Eq. (3) is:

$$\int_{\Omega} \nabla_{\mathbf{X}} \delta \mathbf{u} : \mathbf{P} dV + \int_{\Omega} \rho_0 \delta \mathbf{u} \cdot \ddot{\mathbf{u}} dV = \int_{\Omega} \delta \mathbf{u} \cdot \mathbf{B} dV + \int_{\Gamma^t} \delta \mathbf{u} \cdot \tilde{\mathbf{T}} dA, \quad (5)$$

140 where $\delta \mathbf{u}$ denotes the test function. The function spaces for the trial and test functions are,
 141 respectively:

$$\mathcal{V} = \{ \mathbf{u} \mid \mathbf{u} \in H^1(\Omega), \mathbf{u} = \tilde{\mathbf{u}} \text{ on } \Gamma^u \}, \quad (6a)$$

$$\mathcal{W} = \{ \delta \mathbf{u} \mid \delta \mathbf{u} \in H^1(\Omega), \delta \mathbf{u} = \mathbf{0} \text{ on } \Gamma^u \}, \quad (6b)$$

142 where H^1 is the Sobolev space consisting of functions whose values and first weak derivatives
 143 are square-integrable.

144 We make some assumptions about the problem domain and discretization: The geometry
 145 of the domain Ω can be partitioned into n_{es} non-overlapping identically shaped enrichment
 146 subdomains. Each enrichment subdomain is associated with a unit cell and discretized using a
 147 coarse patch of elements, and the number of coarse-scale finite elements in a patch is denoted

148 by n_{ecp} . The total number of coarse-scale elements is given by $n_{\text{ec}} = n_{\text{es}} \times n_{\text{ecp}}$. The interior
 149 of a subdomain, α , is denoted as Ω_α and its boundary is denoted as Γ_α , with the overall
 150 domain partitioning into subdomains is performed such that $\overline{\Omega} = \bigcup_{\alpha=1}^{n_{\text{es}}} \overline{\Omega}_\alpha$ with $\overline{(\cdot)}$ represents
 151 the closure of (\cdot) . In addition to the coarse-scale discretization, each subdomain is separately
 152 discretized using n_{ef} fine-scale elements that resolve the features of the underlying unit cell.

153 The displacement field over the problem domain is decomposed into coarse and fine-scale
 154 contributions using a two-scale additive decomposition:

$$\mathbf{u} = \mathbf{u}^c + \sum_{\alpha=1}^{n_{\text{es}}} \mathcal{H}(\Omega_\alpha) \mathbf{u}^{f,\alpha}, \quad (7)$$

155 where $\mathcal{H}(\cdot)$ is an indicator function defined below:

$$\mathcal{H}(\Omega_\alpha) = \begin{cases} 1, & \text{for } \mathbf{X} \in \Omega_\alpha \\ 0, & \text{elsewhere.} \end{cases} \quad (8)$$

156 The indicator function in Eq. (8) ensures that only the fine-scale response associated with
 157 subdomain Ω_α contributes to the displacement field within the subdomain. The coarse-scale
 158 field captures the slowly varying component of the solution, whereas the fine-scale fields resolve
 159 the rapidly varying solution due to material heterogeneity. We note that Eq. 7 does not directly
 160 satisfy the continuity condition on \mathbf{u} . The continuity is satisfied by appropriately selecting the
 161 boundary conditions for the fine-scale response field [47]. The test function is decomposed
 162 similarly:

$$\delta \mathbf{u} = \delta \mathbf{u}^c + \sum_{\alpha=1}^{n_{\text{es}}} \mathcal{H}(\Omega_\alpha) \delta \mathbf{u}^{f,\alpha}. \quad (9)$$

163 The finite-dimensional subspaces (following the finite dimensional approximation consistent
 164 with the standard finite element method) for the coarse-scale trial and test functions are
 165 denoted as \mathcal{V}^c and \mathcal{W}^c , respectively, and the corresponding fine-scale function spaces are \mathcal{V}_α^f
 166 and \mathcal{W}_α^f . These spaces are selected such that their direct sum forms the finite-dimensional
 167 subspaces for the original (single-scale) trial and test functions:

$$\mathcal{V}^h = \mathcal{V}^c \oplus \bigoplus_{\alpha=1}^{n_{\text{es}}} \mathcal{V}_\alpha^f, \quad \mathcal{W}^h = \mathcal{W}^c \oplus \bigoplus_{\alpha=1}^{n_{\text{es}}} \mathcal{W}_\alpha^f. \quad (10)$$

168 The fine-scale spaces are defined ($\mathcal{W}_\alpha^f = \mathcal{V}_\alpha^f$) such that the trial and test functions are non-
 169 zero only within the corresponding enrichment subdomain Ω_α and vanish elsewhere except at
 170 the subdomain boundary where the external traction (denoted by Γ_α^t) is applied:

$$\mathcal{V}_\alpha^f \ni \mathbf{u} = \mathbf{0}; \text{ when } \mathbf{X} \in \Gamma_\alpha^u, \quad (11)$$

171 where $\Gamma_\alpha^u = \Gamma_\alpha \setminus \Gamma_\alpha^t$. Equation (11) implies that homogeneous Dirichlet boundary conditions are
 172 imposed at the boundary of each subdomain Ω_α for fine-scale displacement fields, except at Γ_α^t
 173 boundary where the external traction is applied. This boundary condition has been previously
 174 used in similar work [29] for modeling transient dynamical response of phononic crystals and
 175 acoustic metamaterials and in other works [47, 56, 36, 37]. Other boundary conditions (e.g.,
 176 mixed boundary conditions) have also been previously investigated [55]. Moreover, it follows
 177 from the direct sum decomposition given in Eq. (10) that $\mathcal{V}_\alpha^f \cap \mathcal{V}_\beta^f = \emptyset$. Similarly, in order to
 178 ensure direct sum decomposition, the finite-dimensional coarse-scale function spaces $\mathcal{V}^c \subset \mathcal{V}$
 179 and $\mathcal{W}^c \subset \mathcal{W}$ are selected such that:

$$\|\mathbf{u} - \mathbf{v}\|_{\Omega_\alpha} \neq 0; \quad \mathbf{u} \in \mathcal{V}^c; \quad \mathbf{v} \in \mathcal{V}_\alpha^f, \quad (12)$$

180 for any (\mathbf{u}, \mathbf{v}) pair, and $\|\cdot\|_{\Omega_\alpha}$ is L^2 norm over Ω_α .

181 Substituting Eqs. (7) and (9) in Eq. (5), one can decompose the weak form of the linear
 182 momentum balance equation into two tightly coupled problems. The coarse-scale problem is
 183 defined over the entire problem domain as follows:

$$\begin{aligned} \int_{\Omega} \delta \mathbf{u}^c \cdot \rho_0 \ddot{\mathbf{u}}^c dV + \int_{\Omega} \nabla_{\mathbf{X}} \delta \mathbf{u}^c : \mathbf{P}(\mathbf{X}, t, \mathbf{u}^c, \mathbf{u}^{f,\alpha}) dV = \\ - \sum_{\alpha=1}^{n_{es}} \int_{\Omega_\alpha} \delta \mathbf{u}^c \cdot \rho_0 \ddot{\mathbf{u}}^{f,\alpha} dV + \int_{\Omega} \delta \mathbf{u}^c \cdot \mathbf{B} dV + \int_{\Gamma^t} \delta \mathbf{u}^c \cdot \tilde{\mathbf{T}} dA. \end{aligned} \quad (13)$$

184 The terms on the left-hand side of Eq. 13 correspond to kinetic energy and strain energy at the
 185 coarse scale, respectively, while the right-hand side terms correspond to the external work due
 186 to fine-scale dynamics, body force, and traction, respectively. Similarly, the fine-scale problem
 187 in each subdomain Ω_α is obtained as:

$$\begin{aligned} \int_{\Omega_\alpha} \delta \mathbf{u}^{f,\alpha} \cdot \rho_0 \ddot{\mathbf{u}}^{f,\alpha} dV + \int_{\Omega_\alpha} \nabla_{\mathbf{X}} \delta \mathbf{u}^{f,\alpha} : \mathbf{P}(\mathbf{X}, t, \mathbf{u}^c, \mathbf{u}^{f,\alpha}) dV = \\ - \int_{\Omega_\alpha} \delta \mathbf{u}^{f,\alpha} \cdot \rho_0 \ddot{\mathbf{u}}^c dV + \int_{\Omega_\alpha} \delta \mathbf{u}^{f,\alpha} \cdot \mathbf{B} dV + \int_{\Gamma_\alpha^t} \delta \mathbf{u}^{f,\alpha} \cdot \tilde{\mathbf{T}} dA. \end{aligned} \quad (14)$$

188 The terms of Eq. 14 are interpreted in an analogous fashion to the coarse scale problem. The
 189 traction term is only present at the subdomain boundaries that coincide with the exterior
 190 Neumann boundaries of the problem domain, as the fine-scale test function vanishes at all
 191 other subdomain boundaries (see Eq. (11)). The coupling terms in the coarse and fine-scale
 192 problems are highlighted in red, shown respectively in Eqs. (13) and (14).

193 The boundary and initial conditions for the coarse-scale problem and the initial conditions
 194 for the fine-scale problems complete the multiscale governing equations. The initial state at

195 the fine-scale is taken to be undeformed and stationary:

$$\mathbf{u}^{f,\alpha}(\mathbf{X}, 0) = \mathbf{0}, \quad \dot{\mathbf{u}}^{f,\alpha}(\mathbf{X}, 0) = \mathbf{0}, \quad \mathbf{X} \in \Omega_\alpha. \quad (15)$$

196 The boundary and initial conditions for the coarse-scale problem are:

$$\mathbf{u}^c(\mathbf{X}, t) = \tilde{\mathbf{u}}(\mathbf{X}, t), \quad \mathbf{X} \in \Gamma^u; \quad \mathbf{P}(\mathbf{X}, t) \cdot \mathbf{N} = \tilde{\mathbf{T}}(\mathbf{X}, t), \quad \mathbf{X} \in \Gamma^t; \quad (16a)$$

$$\mathbf{u}^c(\mathbf{X}, 0) = \hat{\mathbf{u}}(\mathbf{X}), \quad \dot{\mathbf{u}}^c(\mathbf{X}, 0) = \hat{\mathbf{v}}(\mathbf{X}), \quad \mathbf{X} \in \Omega. \quad (16b)$$

197 The initial displacement ($\hat{\mathbf{u}}$) and velocity ($\hat{\mathbf{v}}$) fields are selected such that they can be accurately
198 described by the coarse-scale discretization.

199 3 Computational Approximation of Two-Scale PDEs

200 This section first describes the spatial and temporal discretization methods used for the mul-
201 tiscale governing equations in the proposed VME formulation. The details of the evaluation
202 of non-standard element matrices and vectors due to multiscale discretization and estimation
203 of stable time increments based on the time integration scheme are discussed later.

204 Numerical simulation of the dynamic response of complex microstructures, particularly
205 in short-wavelength regimes, is often performed using higher-order elements or dissipative
206 time integration schemes. In a related work, spectral elements up to the seventh order were
207 employed for coarse-scale discretization, where each coarse-scale element corresponded to a
208 unit cell [29]. In this study, we employ a strategy that resembles h -refinement by discretizing
209 each unit cell with a patch of coarse-scale elements to accurately capture wave propagation in
210 short-wavelength regimes.

211 3.1 Spatial discretization

212 Consider the decomposition of an enrichment subdomain, Ω_α into a patch of coarse elements:
213 $\Omega_\alpha = \bigcup_{E=1}^{n_{\text{ecp}}} \Delta_{\alpha_E}^c$, where a coarse finite element within that patch is denoted with $\Delta_{\alpha_E}^c$. Using
214 the classical Bubnov-Galerkin approach, the coarse-scale displacement, weighting function,
215 and their gradients for a coarse-scale element are written as:

$$\mathbf{u}_{\alpha_E}^c = \mathbf{N}_{\alpha_E}^c \mathbf{d}_{\alpha_E}^c = \mathbf{N}_{\alpha_E}^c \mathbf{L}_{\alpha_E}^c \mathbf{d}^c, \quad (17a)$$

$$[\nabla \mathbf{u}_{\alpha_E}^c] = \mathbf{B}_{\alpha_E}^c \mathbf{d}_{\alpha_E}^c = \mathbf{B}_{\alpha_E}^c \mathbf{L}_{\alpha_E}^c \mathbf{d}^c, \quad (17b)$$

216 where $\mathbf{u}_{\alpha_E}^c(\mathbf{X}, t) := \mathbf{u}^c(\mathbf{X} \in \Delta_{\alpha_E}^c, t)$. $[\cdot]$ denotes the vectorized form of the tensorial entities.
217 Einstein's summation convention does not apply to the index, α_E . $\mathbf{N}_{\alpha_E}^c(\mathbf{X})$ and $\mathbf{B}_{\alpha_E}^c(\mathbf{X})$

218 are the coarse-scale element shape function matrix and shape function gradient matrix, re-
 219 spectively, for the coarse-scale element, $\Delta_{\alpha_E}^c$. $\mathbf{d}_{\alpha_E}^c(t)$ is the nodal displacement vector for the
 220 coarse-scale element. The local vector is mapped to the corresponding global vector, $\mathbf{d}^c(t)$
 221 through the gather matrix, $\mathbf{L}_{\alpha_E}^c$. The conventional forms typically used in the finite element
 222 literature is employed for these matrices (see e.g., Ref. [15]). The discretizations of the weight
 223 function and its gradient are similarly defined.

224 Each enrichment subdomain is discretized a second time, using n_{ef} fine-scale elements that
 225 resolves the underlying heterogeneous microstructure: $\Omega_\alpha = \bigcup_{e=1}^{n_{\text{ef}}} \Delta_{\alpha_e}^f$. The fine and coarse-
 226 scale discretizations are performed in a compatible fashion; i.e., $\Delta_{\alpha_E}^c = \bigcup_{e \in I_{\alpha_E}} \Delta_{\alpha_e}^f$ for any
 227 macroscale element. I_{α_E} denotes an index set of fine scale elements that resolve the coarse
 228 scale element, α_E . The displacement, weighting function, and their gradients for a fine-scale
 229 element are given as:

$$\mathbf{u}_{\alpha_e}^f = \mathbf{N}_{\alpha_e}^f \mathbf{d}_{\alpha_e}^f = \mathbf{N}_{\alpha_e}^f \mathbf{L}_{\alpha_e}^f \mathbf{d}^{f,\alpha}, \quad (18a)$$

$$[\nabla \mathbf{u}_{\alpha_e}^f] = \mathbf{B}_{\alpha_e}^f \mathbf{d}_{\alpha_e}^f = \mathbf{B}_{\alpha_e}^f \mathbf{L}_{\alpha_e}^f \mathbf{d}^{f,\alpha}, \quad (18b)$$

230 in which, $\mathbf{u}_{\alpha_e}^f$, $\mathbf{N}_{\alpha_e}^f$ and $\mathbf{B}_{\alpha_e}^f$ are defined analogously to their coarse scale counterparts. The
 231 gather matrix, $\mathbf{L}_{\alpha_e}^f$ maps the nodal displacement vector associated with element, α_e (i.e., $\mathbf{d}_{\alpha_e}^f$)
 232 with the nodal displacement vector for the whole enrichment subdomain, $\mathbf{d}^{f,\alpha}$.

233 Using Eqs. (17) and (18), the discretized forms of Eqs. (13) and (14) are obtained as:

$$\mathbf{M}^c \ddot{\mathbf{d}}^c + \sum_{\alpha=1}^{n_{\text{es}}} \mathbf{M}^{cf_\alpha} \ddot{\mathbf{d}}^{f_\alpha} + \mathbf{f}_{\text{int}}^c \left(\mathbf{d}^c, \{ \mathbf{d}^{f,\alpha} \} \right) = \mathbf{f}_{\text{ext}}^c; \quad (19a)$$

$$\mathbf{M}^{f_\alpha} \ddot{\mathbf{d}}^{f_\alpha} + \mathbf{M}^{f_\alpha c} \ddot{\mathbf{d}}^{c,\alpha} + \mathbf{f}_{\text{int}}^{f_\alpha} \left(\mathbf{d}^{c,\alpha}, \mathbf{d}^{f,\alpha} \right) = \mathbf{f}_{\text{ext}}^{f_\alpha}, \quad \alpha = 1 \text{ to } n_{\text{es}}; \quad (19b)$$

234 where $\mathbf{d}^{c,\alpha}$ denotes the nodal displacement vector corresponding to the enrichment subdomain,
 235 Ω_α and related to the total displacement vector through a gather matrix: $\mathbf{d}^{c,\alpha} = \mathbf{L}_\alpha^c \mathbf{d}^c$. \mathbf{M}^c
 236 and \mathbf{M}^{f_α} are the coarse- and fine-scale mass matrices, respectively; \mathbf{M}^{cf_α} and $\mathbf{M}^{f_\alpha c}$ are the
 237 mass matrices that describe scale interactions. The internal force vectors at the coarse and
 238 fine-scales are given by $\mathbf{f}_{\text{int}}^c$, $\mathbf{f}_{\text{int}}^{f_\alpha}$ respectively, and the external force vectors at coarse and
 239 fine-scale are $\mathbf{f}_{\text{ext}}^c$ and $\mathbf{f}_{\text{ext}}^{f_\alpha}$, respectively. These are obtained by assembling the element-level

240 matrices and internal and external force vectors as shown below:

$$\mathbf{M}^c = \sum_{\alpha, E} (\mathbf{L}_{\alpha_E}^c)^T \mathbf{M}_{\alpha_E}^c \mathbf{L}_{\alpha_E}^c; \quad \mathbf{M}^{cf_\alpha} = \left(\mathbf{M}^{f_\alpha c} \right)^T = \sum_{E, e \in I_{\alpha_E}} (\mathbf{L}_{\alpha_E}^c)^T \mathbf{M}_{E, e}^\alpha \mathbf{L}_{\alpha_e}^f; \quad (20a)$$

$$\mathbf{M}^{f_\alpha} = \sum_e (\mathbf{L}_{\alpha_e}^f)^T \mathbf{M}_e^{f_\alpha} \mathbf{L}_{\alpha_e}^f; \quad \mathbf{f}_{int}^c = \sum_{\alpha, E} (\mathbf{L}_{\alpha_E}^c)^T \mathbf{f}_{int}^{c, \alpha_E}; \quad (20b)$$

$$\mathbf{f}_{int}^{f_\alpha} = \sum_e (\mathbf{L}_{\alpha_e}^f)^T \mathbf{f}_{int}^{f_\alpha, e}; \quad \mathbf{f}_{ext}^c = \sum_{\alpha, E} (\mathbf{L}_{\alpha_E}^c)^T \mathbf{f}_{ext}^{c, \alpha_E}; \quad \mathbf{f}_{ext}^{f_\alpha} = \sum_e (\mathbf{L}_{\alpha_e}^f)^T \mathbf{f}_{ext}^{f_\alpha, e}. \quad (20c)$$

241 The corresponding coarse-scale element-level matrices and force vectors are obtained as:

$$\mathbf{M}_{\alpha_E}^c = \int_{\Omega_{\alpha_E}} (\mathbf{N}_{\alpha_E}^c)^T \rho_0 \mathbf{N}_{\alpha_E}^c dV; \quad \mathbf{f}_{int}^{c, \alpha_E} = \int_{\Omega_{\alpha_E}} (\mathbf{B}_{\alpha_E}^c)^T [\mathbf{P}] dV; \quad (21a)$$

$$\mathbf{f}_{ext}^{c, \alpha_E} = \int_{\Omega_{\alpha_E}} (\mathbf{N}_{\alpha_E}^c)^T \mathbf{B} dV + \int_{\Gamma^t} (\mathbf{N}_{\alpha_E}^c)^T \tilde{\mathbf{T}} dA; \quad (21b)$$

242 where $[\mathbf{P}]$ is the vectorized form of the first Piola-Kirchhoff stress tensor. The vectorized form
243 (distinct from the Voigt notation) includes all components of the stress tensor since it is not
244 symmetric. In the case of a composite microstructure where the density spatially varies, the
245 integrations for $\mathbf{M}_{\alpha_E}^c$ and $\mathbf{f}_{ext}^{c, \alpha_E}$ are performed by further discretizing the coarse-scale elements
246 into its fine-scale counterparts or by approximating the integration by averaging the density.
247 The integration for the internal force vector is performed over the underlying fine-scale grid
248 to capture the stress variations within the microstructure.

249 The fine-scale element-level matrices and force vectors are obtained as:

$$\mathbf{M}_e^{f_\alpha} = \int_{\Omega_{\alpha_e}} (\mathbf{N}_{\alpha_e}^f)^T \rho_0 \mathbf{N}_{\alpha_e}^f dV; \quad \mathbf{M}_{E, e}^\alpha = \int_{\Omega_{\alpha_e}} (\mathbf{N}_{\alpha_E}^c)^T \rho_0 \mathbf{N}_{\alpha_e}^f dV; \quad (22a)$$

$$\mathbf{f}_{int}^{f_\alpha, e} = \int_{\Omega_{\alpha_e}} (\mathbf{B}_{\alpha_e}^f)^T [\mathbf{P}] dV; \quad \mathbf{f}_{ext}^{f_\alpha, e} = \int_{\Omega_{\alpha_e}} (\mathbf{N}_{\alpha_e}^f)^T \mathbf{B} dV + \int_{\Gamma_{\alpha_e}^t} (\mathbf{N}_{\alpha_e}^f)^T \tilde{\mathbf{T}} dA. \quad (22b)$$

250 The evaluation of $\mathbf{M}_{\alpha_E}^c$ and $\mathbf{M}_{E, e}^\alpha$, $\mathbf{f}_{int}^{c, \alpha_E}$ and $\mathbf{f}_{int}^{f_\alpha, e}$ is non-standard and further discussed in
251 Section 3.4. The remaining entities in Eqs. (21) and (22) are evaluated using the standard
252 element-level integration procedure.

253 3.2 Time integration

254 The coupled two-scale semi-discrete multiscale equations given in Eq. (19) are integrated it-
255 eratively using an operator-split procedure until convergence is obtained for each time step.
256 Within the operator-split procedure, various time-integration schemes for the coarse-scale and
257 fine-scale equations are employed in this work, namely: (1) explicit update for both equa-
258 tions using the central difference method, (2) explicit Noh and Bathe [43] scheme for both

equations, and (3) mixed explicit-implicit time integration scheme with explicit integration for the coarse-scale equations using Noh and Bathe [43] scheme and implicit integration for the fine-scale equations using Bathe and Baig [5] scheme. The performance of these integration schemes is assessed in Section 4.4.

The following notation is adopted: Consider that the global displacement, velocity, and acceleration vectors at the coarse and fine scales are known at time, t_n (also referred to as the n^{th} time step). These vectors are denoted by $(\mathbf{d}_n^c, \dot{\mathbf{d}}_n^c, \ddot{\mathbf{d}}_n^c)$ and $(\mathbf{d}_n^{f,\alpha}, \dot{\mathbf{d}}_n^{f,\alpha}, \ddot{\mathbf{d}}_n^{f,\alpha})$ respectively. The time integration results in the corresponding coarse and fine-scale fields at the $(n + 1)^{\text{th}}$ time step, which are denoted as $(\mathbf{d}_{n+1}^c, \dot{\mathbf{d}}_{n+1}^c, \ddot{\mathbf{d}}_{n+1}^c)$ and $(\mathbf{d}_{n+1}^{f,\alpha}, \dot{\mathbf{d}}_{n+1}^{f,\alpha}, \ddot{\mathbf{d}}_{n+1}^{f,\alpha})$ respectively. The state of a vector at the k^{th} iteration of the operator-split procedure within $[t_n, t_{n+1}]$ is denoted by a second subscript (e.g., $\ddot{\mathbf{d}}_{n+1,k}^c$).

3.2.1 Explicit-explicit central difference method (EE-CDM)

In this method, both the coarse- and the fine-scale equations (Eqs. (19)a-b) are integrated using the explicit central difference method. The implementation procedure is provided in Algorithm 1. At a given time increment, t_n , the algorithm updates the multiscale nodal displacement vectors, first. The acceleration vectors are computed iteratively until convergence. Convergence is assessed based on discrete L^∞ norm, in order to ensure that the accuracy tolerance is strictly enforced component-wise. Upon convergence, the velocity vectors are updated. EE-CDM is the most straightforward and efficient method among the three investigated, but, similar to the single-scale implementation, the multiscale implementation proposed here also could suffer from instability.

3.2.2 Explicit-explicit sub-step method (EE-SSM)

Despite its large time step stability limit among explicit schemes, it is well-known that dispersion errors in high-frequency modes can lead to degraded solution accuracy when the central difference method is employed. The integration procedure for EE-SSM, which is used to alleviate spurious high frequency dispersion, is described in Algorithm 2. The coarse and fine-scale problems are integrated using an explicit sub-step time integration scheme proposed by Noh and Bathe [43]. In this scheme, the coarse and fine-scale fields are integrated in two steps, first at a sub-step denoted by t_{n+p} ($0 < p < 1$) and then at the full step denoted by t_{n+1} . In both updates, the operator-split procedure is used to iteratively solve for coarse- and fine-scale fields.

Algorithm 1 Implementation of EE-CDM.

- 1: **Input:** Given the state at t_n and the external force vectors at t_{n+1} .
- 2: Estimate a stable time increment, Δt , for the multiscale problem as described in Section 3.5. Set $t_{n+1} = t_n + \Delta t$ and iteration count $k = 1$.
- 3: Set the initial guess for the fine-scale acceleration: $\ddot{\mathbf{d}}_{n+1,0}^{\text{f},\alpha} = \ddot{\mathbf{d}}_n^{\text{f},\alpha}$.
- 4: Update the displacement fields:

$$\begin{aligned}\mathbf{d}_{n+1}^{\text{f},\alpha} &= \mathbf{d}_n^{\text{f},\alpha} + \Delta t \dot{\mathbf{d}}_n^{\text{f},\alpha} + \frac{(\Delta t)^2}{2} \ddot{\mathbf{d}}_n^{\text{f},\alpha}; \quad \alpha = 1, 2, \dots, n_{\text{es}}, \\ \mathbf{d}_{n+1}^{\text{c}} &= \mathbf{d}_n^{\text{c}} + \Delta t \dot{\mathbf{d}}_n^{\text{c}} + \frac{(\Delta t)^2}{2} \ddot{\mathbf{d}}_n^{\text{c}}.\end{aligned}$$

- 5: At any arbitrary iteration count, k :
- a: Explicitly integrate the coarse-scale acceleration:

$$\mathbf{M}^{\text{c}} \ddot{\mathbf{d}}_{n+1,k}^{\text{c}} + \sum_{\alpha=1}^{n_{\text{ec}}} \mathbf{M}^{\text{f},\alpha} \ddot{\mathbf{d}}_{n+1,k-1}^{\text{f},\alpha} + \mathbf{f}_{\text{int}}^{\text{c}} \left(\mathbf{d}_{n+1}^{\text{c}}, \mathbf{d}_{n+1}^{\text{f}} \right) = \mathbf{f}_{\text{ext}}^{\text{c}}.$$

- b: Explicitly integrate the fine-scale acceleration for each subdomain, α :

$$\mathbf{M}^{\text{f},\alpha} \ddot{\mathbf{d}}_{n+1,k}^{\text{f},\alpha} + \mathbf{M}^{\text{f},\alpha} \mathbf{L}_{\alpha}^{\text{c}} \ddot{\mathbf{d}}_{n+1,k}^{\text{c}} + \mathbf{f}_{\text{int}}^{\text{f},\alpha} \left(\mathbf{d}_{n+1}^{\text{c},\alpha}, \mathbf{d}_{n+1}^{\text{f},\alpha} \right) = \mathbf{f}_{\text{ext}}^{\text{f},\alpha}.$$

- c: Check for convergence :

$$e_{n+1}^{\text{f},\alpha} := \frac{\|\ddot{\mathbf{d}}_{n+1,k}^{\text{f},\alpha} - \ddot{\mathbf{d}}_{n+1,k-1}^{\text{f},\alpha}\|_{\infty}}{\|\ddot{\mathbf{d}}_{n+1,k-1}^{\text{f},\alpha}\|_{\infty}} < \text{tol}_{\text{f}}; \quad e_{n+1}^{\text{c}} := \frac{\|\ddot{\mathbf{d}}_{n+1,k}^{\text{c}} - \ddot{\mathbf{d}}_{n+1,k-1}^{\text{c}}\|_{\infty}}{\|\ddot{\mathbf{d}}_{n+1,k-1}^{\text{c}}\|_{\infty}} < \text{tol}_{\text{c}}.$$

- 6: **If** 5.c is met **then** $\ddot{\mathbf{d}}_{n+1}^{\text{f},\alpha} \leftarrow \ddot{\mathbf{d}}_{n+1,k}^{\text{f},\alpha}$ and $\ddot{\mathbf{d}}_{n+1}^{\text{c}} \leftarrow \ddot{\mathbf{d}}_{n+1,k}^{\text{c}}$.
- 7: **Else** $k \leftarrow k + 1$, and **go to** 5 for the next iteration.
- 8: Update the velocity fields:

$$\begin{aligned}\dot{\mathbf{d}}_{n+1}^{\text{f},\alpha} &= \dot{\mathbf{d}}_n^{\text{f},\alpha} + \frac{\Delta t}{2} \left(\ddot{\mathbf{d}}_n^{\text{f},\alpha} + \ddot{\mathbf{d}}_{n+1}^{\text{f},\alpha} \right); \quad \alpha = 1, 2, \dots, n_{\text{es}}, \\ \dot{\mathbf{d}}_{n+1}^{\text{c}} &= \dot{\mathbf{d}}_n^{\text{c}} + \frac{\Delta t}{2} \left(\ddot{\mathbf{d}}_n^{\text{c}} + \ddot{\mathbf{d}}_{n+1}^{\text{c}} \right).\end{aligned}$$

- 9: **return** $(\mathbf{d}_{n+1}^{\text{c}}, \dot{\mathbf{d}}_{n+1}^{\text{c}}, \ddot{\mathbf{d}}_{n+1}^{\text{c}})$ and $(\mathbf{d}_{n+1}^{\text{f},\alpha}, \dot{\mathbf{d}}_{n+1}^{\text{f},\alpha}, \ddot{\mathbf{d}}_{n+1}^{\text{f},\alpha})$.

290

3.2.3 Explicit-implicit sub-step method (EI-SSM)

291

We also explore an explicit-implicit sub-step time integration method, where the coarse-scale problem is integrated explicitly using the Noh and Bathe [43] scheme and the fine-scale problems are implicitly integrated using the Bathe and Baig [5] scheme. The resulting multiscale scheme is not unconditionally stable as the coarse-scale equation is updated explicitly, but the stable time increment of the coarse-scale problem is expected to be larger than that of

Algorithm 2 Implementation of EE-SSM.

- 1: **Input:** Given the state at t_n and the external force vectors at t_{n+1} .
- 2: Estimate a stable time increment, Δt , and set the sub-step ratio p . $t_{n+p} = t_n + p\Delta t$ and iteration count $k = 1$. Set the constants of integration:

$$\begin{aligned} q_1 &= \frac{1-2p}{2p(1-p)}; & q_2 &= \frac{1}{2} - pq_1; & q_0 &= -q_1 - q_2 + \frac{1}{2}; & a_0 &= p\Delta t; & a_1 &= \frac{1}{2}(p\Delta t)^2; \\ a_2 &= a_0/2; & a_3 &= (1-p)\Delta t; & a_4 &= \frac{1}{2}a_3^2; & a_5 &= q_0a_3; & a_6 &= (0.5 + q_1)a_3; & a_7 &= q_2a_3. \end{aligned}$$

Sub-step:

- 3: Compute the external force vectors at t_{n+p} .
- 4: Set the initial guess for the fine-scale acceleration: $\ddot{\mathbf{d}}_{n+p,0}^{\text{f},\alpha} = \ddot{\mathbf{d}}_n^{\text{f},\alpha}$.
- 5: Update the fine- and coarse-scale displacement fields:

$$\mathbf{d}_{n+p}^{\text{f},\alpha} = \mathbf{d}_n^{\text{f},\alpha} + a_0 \dot{\mathbf{d}}_n^{\text{f},\alpha} + a_1 \ddot{\mathbf{d}}_n^{\text{f},\alpha}; \quad \mathbf{d}_{n+p}^{\text{c}} = \mathbf{d}_n^{\text{c}} + a_0 \dot{\mathbf{d}}_n^{\text{c}} + a_1 \ddot{\mathbf{d}}_n^{\text{c}}.$$

- 6: At an arbitrary increment, $k > 0$:

- a: Explicitly integrate the coarse- and fine-scale accelerations:

$$\begin{aligned} \mathbf{M}^{\text{c}} \ddot{\mathbf{d}}_{n+p,k}^{\text{c}} + \sum_{\alpha=1}^{n_{\text{es}}} \mathbf{M}^{\text{cf},\alpha} \ddot{\mathbf{d}}_{n+p,k-1}^{\text{f},\alpha} + \mathbf{f}_{\text{int}}^{\text{c}} \left(\mathbf{d}_{n+p}^{\text{c}}, \mathbf{d}_{n+p}^{\text{f}} \right) &= \mathbf{f}_{\text{ext}}^{\text{c}}, \\ \mathbf{M}^{\text{f},\alpha} \ddot{\mathbf{d}}_{n+p,k}^{\text{f},\alpha} + \mathbf{M}^{\text{f},\alpha} \mathbf{L}_{\alpha}^{\text{c}} \ddot{\mathbf{d}}_{n+p,k}^{\text{c}} + \mathbf{f}_{\text{int}}^{\text{f},\alpha} \left(\mathbf{d}_{n+p}^{\text{c},\alpha}, \mathbf{d}_{n+p}^{\text{f},\alpha} \right) &= \mathbf{f}_{\text{ext}}^{\text{f},\alpha}. \end{aligned}$$

- b: Check for convergence:

$$e_{n+p}^{\text{f},\alpha} < \text{tol}_f; \quad e_{n+p}^{\text{c}} < \text{tol}_c.$$

- 7: **If** 6.b is met **then** $\ddot{\mathbf{d}}_{n+p}^{\text{f},\alpha} \leftarrow \ddot{\mathbf{d}}_{n+p,k}^{\text{f},\alpha}$ and $\ddot{\mathbf{d}}_{n+p}^{\text{c}} \leftarrow \ddot{\mathbf{d}}_{n+p,k}^{\text{c}}$.

- 8: **Else** $k \leftarrow k + 1$, and **go to** 6 for the next iteration.

- 9: Update the velocity fields:

$$\dot{\mathbf{d}}_{n+p}^{\text{f},\alpha} = \dot{\mathbf{d}}_n^{\text{f},\alpha} + a_2(\ddot{\mathbf{d}}_n^{\text{f},\alpha} + \ddot{\mathbf{d}}_{n+p}^{\text{f},\alpha}); \quad \dot{\mathbf{d}}_{n+p}^{\text{c}} = \dot{\mathbf{d}}_n^{\text{c}} + a_2(\ddot{\mathbf{d}}_n^{\text{c}} + \ddot{\mathbf{d}}_{n+p}^{\text{c}}).$$

Full-step:

- 10: Set $k = 1$; Compute the external force vectors at t_{n+1} and set the initial guess for the fine-scale acceleration: $\ddot{\mathbf{d}}_{n+1,0}^{\text{f},\alpha} = \ddot{\mathbf{d}}_{n+p}^{\text{f},\alpha}$.
- 11: Update the fine- and coarse-scale displacement fields:

$$\mathbf{d}_{n+1}^{\text{f},\alpha} = \mathbf{d}_{n+p}^{\text{f},\alpha} + a_3 \dot{\mathbf{d}}_{n+p}^{\text{f},\alpha} + a_4 \ddot{\mathbf{d}}_{n+p}^{\text{f},\alpha}; \quad \mathbf{d}_{n+1}^{\text{c}} = \mathbf{d}_{n+p}^{\text{c}} + a_3 \dot{\mathbf{d}}_{n+p}^{\text{c}} + a_4 \ddot{\mathbf{d}}_{n+p}^{\text{c}}.$$

- 12: Iterate the coarse- and fine-scale accelerations until convergence by following step 5 of the Algorithm 1 provided for the central difference method.

- 13: Update the velocity fields:

$$\dot{\mathbf{d}}_{n+1}^{\text{f},\alpha} = \dot{\mathbf{d}}_{n+p}^{\text{f},\alpha} + a_5 \ddot{\mathbf{d}}_n^{\text{f},\alpha} + a_6 \ddot{\mathbf{d}}_{n+p}^{\text{f},\alpha} + a_7 \ddot{\mathbf{d}}_{n+1}^{\text{f},\alpha}; \quad \dot{\mathbf{d}}_{n+1}^{\text{c}} = \dot{\mathbf{d}}_{n+p}^{\text{c}} + a_5 \ddot{\mathbf{d}}_n^{\text{c}} + a_6 \ddot{\mathbf{d}}_{n+p}^{\text{c}} + a_7 \ddot{\mathbf{d}}_{n+1}^{\text{c}}.$$

- 14: **return** $(\mathbf{d}_{n+1}^{\text{c}}, \dot{\mathbf{d}}_{n+1}^{\text{c}}, \ddot{\mathbf{d}}_{n+1}^{\text{c}})$ and $(\mathbf{d}_{n+1}^{\text{f},\alpha}, \dot{\mathbf{d}}_{n+1}^{\text{f},\alpha}, \ddot{\mathbf{d}}_{n+1}^{\text{f},\alpha})$.

296 fine-scale problems. Hence, the time increment for this scheme is expected to be governed by
 297 accuracy requirements rather than the stability constraints of the fine-scale equations. This
 298 approach could be advantageous compared to the explicit-explicit sub-step method, if the time
 299 step required to ensure accuracy is sufficiently larger than the stability limit such that the over-
 300 all integration method is computationally beneficial despite the added computational cost of
 301 the implicit update. A microstructure with high stiffness contrast, where the stability of the
 302 fine-scale explicit update severely constrains the time step size, is such a problem.

303 As the model exhibits geometric and material nonlinearity, the Newton-Raphson method is
 304 employed to solve the nonlinear fine-scale equations in EI-SSM. To solve the fine-scale equations
 305 using the Newton-Raphson method, the first variation of the residual given in Eq. (14) is taken
 306 along the direction $(\tilde{d}\mathbf{u}^{f,\alpha}, \tilde{d}\ddot{\mathbf{u}}^{f,\alpha})$ to obtain the corresponding Jacobian. The incremental fine-
 307 scale fields are discretized analogously as described in Eq. (18) to obtain $(\tilde{d}\mathbf{d}^{f,\alpha}, \tilde{d}\ddot{\mathbf{d}}^{f,\alpha})$ and
 308 using the classical Bubnov-Galerkin approach, the semi-discrete version of the Jacobian is
 309 obtained. The semi-discrete Jacobian and the semi-discrete residual (given in Eq. (19)b) form
 310 the linearized system of equations for the incremental fine-scale fields as shown below:

$$\mathbf{M}^{f_\alpha} \tilde{d}\ddot{\mathbf{d}}^{f,\alpha} + \mathbf{K}^{f_\alpha}(\mathbf{d}^{c,\alpha}, \mathbf{d}^{f,\alpha}) \tilde{d}\mathbf{d}^{f,\alpha} = - \left(\mathbf{M}^{f_\alpha} \tilde{d}\ddot{\mathbf{d}}^{f,\alpha} + \mathbf{f}_{int}^{f_\alpha}(\mathbf{d}^{c,\alpha}, \mathbf{d}^{f,\alpha}) + \mathbf{M}^{f_\alpha c} \tilde{d}\ddot{\mathbf{d}}^c - \mathbf{f}_{ext}^{f_\alpha} \right), \quad (23)$$

311 where \mathbf{K}^{f_α} is the fine-scale tangent stiffness matrix. It can be obtained using the corresponding
 312 element-level matrix as shown below:

$$\mathbf{K}^{f_\alpha} = \sum_e \left(\mathbf{L}_{\alpha_e}^f \right)^T \mathbf{K}_e^{f_\alpha} \mathbf{L}_{\alpha_e}^f, \quad \text{where} \quad \mathbf{K}_e^{f_\alpha} = \int_{\Omega_{\alpha_e}} \left(\mathbf{B}_{\alpha_e}^f \right)^T [\mathbf{D}] \mathbf{B}_{\alpha_e}^f dV. \quad (24)$$

313 The $[\mathbf{D}]$ is the matrix form of the fourth-order tensor $\partial \mathbf{P} / \partial \mathbf{F} (\mathbf{d}^{c,\alpha}, \mathbf{d}^{f,\alpha})$. The fully-discrete
 314 version of the linearized system of fine-scale fields is obtained by substituting the incremental
 315 acceleration in terms of incremental displacement in Eq. (23). To do the same, the implicit
 316 acceleration update is linearized in terms of the unknown displacement field, both at the sub-
 317 step and full-step. For the sub-step, the following linear system of equations is obtained which
 318 is solved iteratively using the Newton-Raphson (N-R) method:

$$\begin{aligned} & \left(\frac{4}{p^2 \Delta t^2} \mathbf{M}^{f_\alpha} + \mathbf{K}^{f_\alpha} \left(\mathbf{d}_{n+p}^{c,\alpha}, (\mathbf{d}_{n+p,k}^{f,\alpha})^{(i-1)} \right) \right) (\tilde{d}\mathbf{d}_{n+p,k}^{f,\alpha})^{(i)} = \\ & - \mathbf{M}^{f_\alpha} (\tilde{d}\ddot{\mathbf{d}}_{n+p,k}^{f,\alpha})^{(i-1)} - \mathbf{f}_{int}^{f_\alpha} \left(\mathbf{d}_{n+p}^{c,\alpha}, (\mathbf{d}_{n+p,k}^{f,\alpha})^{(i-1)} \right) - \mathbf{M}^{f_\alpha c} \tilde{d}\ddot{\mathbf{d}}_{n+p,k}^c + \mathbf{f}_{ext}^{f_\alpha}, \end{aligned} \quad (25)$$

319 where (i) superscript corresponds to the vectorial fields at the i^{th} iteration of the N-R method.

Similarly, the following linear system of equations is obtained at the full-step:

$$\begin{aligned} & \left(c_3 c_3 \mathbf{M}^{f_\alpha} + \mathbf{K}^{f_\alpha} \left(\mathbf{d}_{n+1}^{c,\alpha}, (\mathbf{d}_{n+1,k}^{f,\alpha})^{(i-1)} \right) \right) (\tilde{\mathbf{d}} \mathbf{d}_{n+1,k}^{f,\alpha})^{(i)} = \\ & - \mathbf{M}^{f_\alpha} (\ddot{\mathbf{d}}_{n+1,k}^{f,\alpha})^{(i-1)} - \mathbf{f}_{\text{int}}^{f_\alpha} \left(\mathbf{d}_{n+1}^{c,\alpha}, (\mathbf{d}_{n+1,k}^{f,\alpha})^{(i-1)} \right) - \mathbf{M}^{f_\alpha c} \ddot{\mathbf{d}}_{n+1,k}^c + \mathbf{f}_{\text{ext}}^{f_\alpha}, \end{aligned} \quad (26)$$

321 which can be similarly solved using N-R method.

322 The implicit update of the fine-scale problem for the sub-step and full-step are described
323 in Algorithm 3. The coarse-scale problem update remains the same as discussed in Algorithm
324 2. The sub-step ratio, p , is taken to be the same for both coarse and fine-scale problems for
325 consistent evaluation of fields for both scales at the sub-step and full-step. Unlike EE-CDM
and EE-SSM, the stable time increment is chosen based on the coarse-scale problem.

Algorithm 3 Implementation of EI-SSM.

- 1: **Input:** Given the state at t_n and the external force vectors at t_{n+1} .
- 2: Estimate a stable time increment, Δt , and set the sub-step ratio p . $t_{n+p} = t_n + p\Delta t$ and iteration count $k = 1$. Get the integration constants for the implicit scheme as given below:

$$c_1 = (1 - p)/(p \Delta t); \quad c_2 = -1/((1 - p)p \Delta t); \quad c_3 = (2 - p)/((1 - p)\Delta t).$$

Sub-step:

- 3: Compute the external force vector at t_{n+p} ; and set the initial guesses: $\mathbf{d}_{n+p,0}^{f,\alpha} = \mathbf{d}_n^{f,\alpha}$, $\ddot{\mathbf{d}}_{n+p,0}^{f,\alpha} = \ddot{\mathbf{d}}_n^{f,\alpha}$.
- 4: At an arbitrary increment, $k > 0$:
 - a: The coarse-scale acceleration vector is updated as given in step 5 of Algorithm 2.
 - b: Solve Eq.(25) iteratively for the fine-scale displacement and acceleration corresponding to each subdomain, α , and the updates for the i^{th} N-R iteration are shown below:

$$\begin{aligned} (\mathbf{d}_{n+p,k}^{f,\alpha})^{(i)} &= (\mathbf{d}_{n+p,k}^{f,\alpha})^{(i-1)} + (\tilde{\mathbf{d}} \mathbf{d}_{n+p,k}^{f,\alpha})^{(i)}; \\ (\ddot{\mathbf{d}}_{n+p,k}^{f,\alpha})^{(i)} &= \left((\mathbf{d}_{n+p,k}^{f,\alpha})^{(i)} - \mathbf{d}_n^{f,\alpha} - \dot{\mathbf{d}}_n^{f,\alpha} p \Delta t \right) \frac{4}{p^2 \Delta t^2} - \ddot{\mathbf{d}}_n^{f,\alpha}. \end{aligned}$$

The Newton iterations are performed until the norm of the discrete residual vector (the right-hand side of the Eq. (25)) is below a set tolerance value.

- c: Check the following error quantities in addition to those mentioned in step 6.b of Algorithm 2:

$$\frac{\|\mathbf{d}_{n+p,k}^{f,\alpha} - \mathbf{d}_{n+p,k-1}^{f,\alpha}\|_\infty}{\|\mathbf{d}_{n+p,k-1}^{f,\alpha}\|_\infty} < \text{tol}_f.$$

- 5: If 4.c is met, then $\mathbf{d}_{n+p}^{f,\alpha} \leftarrow \mathbf{d}_{n+p,k}^{f,\alpha}$, $\ddot{\mathbf{d}}_{n+p}^{f,\alpha} \leftarrow \ddot{\mathbf{d}}_{n+p,k}^{f,\alpha}$, and $\ddot{\mathbf{d}}_{n+p}^c \leftarrow \ddot{\mathbf{d}}_{n+p,k}^c$.
- 6: Else $k \leftarrow k + 1$, and go to 4 for the next iteration.
- 7: The coarse-scale velocity is updated as in step 9 of Algorithm 2, and the fine-scale velocity update is:

$$\dot{\mathbf{d}}_{n+p}^{f,\alpha} = \left(\mathbf{d}_{n+p}^{f,\alpha} - \mathbf{d}_n^{f,\alpha} \right) \frac{2}{p \Delta t} - \dot{\mathbf{d}}_n^{f,\alpha}; \quad \alpha = 1, 2, \dots, n_{\text{es}}.$$

Algorithm 3 Implementation of EI-SSM (cont.)

8: **Full-step:**

9: Set $k = 1$; Compute the external force vectors at t_{n+1} and set the initial guesses: $\mathbf{d}_{n+1,0}^{f,\alpha} = \mathbf{d}_{n+p}^{f,\alpha}$, $\ddot{\mathbf{d}}_{n+1,0}^{f,\alpha} = \ddot{\mathbf{d}}_{n+p}^{f,\alpha}$.

10: At an arbitrary increment, $k > 0$:

a: Update the coarse-scale accelerations as given in step 11 of Algorithm 2.

b: Solve Eq.(26) iteratively for fine-scale displacement and acceleration, and the updates for the i^{th} iteration are shown below:

$$(\mathbf{d}_{n+1,k}^{f,\alpha})^{(i)} = (\mathbf{d}_{n+1,k}^{f,\alpha})^{(i-1)} + (\tilde{d}\mathbf{d}_{n+1,k}^{f,\alpha})^{(i)},$$

$$(\ddot{\mathbf{d}}_{n+1,k}^{f,\alpha})^{(i)} = c_3 \left(c_3 (\mathbf{d}_{n+1,k}^{f,\alpha})^{(i)} + c_2 \mathbf{d}_{n+p}^{f,\alpha} + c_1 \mathbf{d}_n^{f,\alpha} \right) + c_2 \dot{\mathbf{d}}_{n+p}^{f,\alpha} + c_1 \dot{\mathbf{d}}_n^{f,\alpha}.$$

The Newton iterations are performed until the norm of the residual vector (the right-hand side of the Eq. (26)) is below a set tolerance value.

c: Check the following error quantities in addition to those mentioned in step 5.c of Algorithm 1:

$$\frac{\|\mathbf{d}_{n+1,k}^{f,\alpha} - \mathbf{d}_{n+1,k-1}^{f,\alpha}\|_{\infty}}{\|\mathbf{d}_{n+1,k-1}^{f,\alpha}\|_{\infty}} < \text{tol}_f; \quad \alpha = 1, 2, \dots, n_{\text{es}}.$$

11: **If** 10.c is met, **then** $\mathbf{d}_{n+1}^{f,\alpha} \leftarrow \mathbf{d}_{n+1,k}^{f,\alpha}$, $\ddot{\mathbf{d}}_{n+1}^{f,\alpha} \leftarrow \ddot{\mathbf{d}}_{n+1,k}^{f,\alpha}$, and $\ddot{\mathbf{d}}_{n+1}^c \leftarrow \ddot{\mathbf{d}}_{n+1,k}^c$.

12: **Else** $k \leftarrow k + 1$, and **go to** 10 for the next iteration.

13: The coarse-scale velocity is updated as in step 13 of Algorithm 2 and the fine-scale velocity update is:

$$\dot{\mathbf{d}}_{n+1}^{f,\alpha} = c_3 \mathbf{d}_{n+1}^{f,\alpha} + c_2 \mathbf{d}_{n+p}^{f,\alpha} + c_1 \mathbf{d}_n^{f,\alpha}.$$

14: **return** $(\mathbf{d}_{n+1}^c, \dot{\mathbf{d}}_{n+1}^c, \ddot{\mathbf{d}}_{n+1}^c)$ and $(\mathbf{d}_{n+1}^{f,\alpha}, \dot{\mathbf{d}}_{n+1}^{f,\alpha}, \ddot{\mathbf{d}}_{n+1}^{f,\alpha})$.

327

3.3 Overall algorithm

328

The overall algorithm for the semi-discrete multiscale equations is described in Algorithm 4.

329

Given the initial displacement and velocity conditions at coarse- and fine-scales, the initial 330 accelerations at both scales are obtained iteratively until convergence is achieved. For an arbitrary 331 time step, the stable time increment is obtained based on the time integration method, 332 and appropriate updates to the coarse- and fine-scale fields are performed.

333

3.4 Element matrices and vectors

334

The evaluation of element-level matrices and vectors given in Eqs. (21), (22) requires numerical 335 integration of the appropriate entities that involve coarse-scale basis functions only, fine-scale 336 basis functions only, and some involving both coarse and fine-scale basis functions. For brevity, 337 the non-standard integration procedure for the internal force vector ($\mathbf{f}_{\text{int}}^{c,\alpha_E}$) in the coarse- 338 scale problem is discussed below, and other non-standard element-level entities are evaluated

Algorithm 4 Algorithm for multiscale problem

- 1: **Input:** Given an initial displacement and velocity condition for the coarse-scale problem $(\mathbf{d}_0^c, \dot{\mathbf{d}}_0^c)$, along with loading and boundary conditions. The initial fine-scale displacement and velocity are $(\mathbf{d}_0^{f,\alpha} = \mathbf{0}, \dot{\mathbf{d}}_0^{f,\alpha} = \mathbf{0})$.
- 2: **Initial acceleration:** The initial acceleration for coarse-scale $(\ddot{\mathbf{d}}_0^c)$ and fine-scale $(\ddot{\mathbf{d}}_0^{f,\alpha})$ problems are obtained from iteratively solving Eq. (19a) and Eq. (19b), until convergence is achieved. Set $n = 0$.
- 3: Update for $(n + 1)^{\text{th}}$ time step:
 - a: Known fields at n^{th} time step: coarse-scale $(\mathbf{d}_n^c, \dot{\mathbf{d}}_n^c, \ddot{\mathbf{d}}_n^c)$ and fine-scale $(\mathbf{d}_n^{f,\alpha}, \dot{\mathbf{d}}_n^{f,\alpha}, \ddot{\mathbf{d}}_n^{f,\alpha})$.
 - b: Get a stable time increment (Δt) for the multiscale problem as discussed in Sec. 3.5, which depends on the time integration method being employed.
 - c: The coarse-scale $(\mathbf{d}_{n+1}^c, \dot{\mathbf{d}}_{n+1}^c, \ddot{\mathbf{d}}_{n+1}^c)$ and fine-scale $(\mathbf{d}_{n+1}^{f,\alpha}, \dot{\mathbf{d}}_{n+1}^{f,\alpha}, \ddot{\mathbf{d}}_{n+1}^{f,\alpha})$ using Algorithm 1 for *EE-CDM*, Algorithm 2 for *EE-SSM*, and Algorithm 3 for *EI-SSM*.
- 4: Set $n \leftarrow n + 1$, and repeat Step 3 until desired.

339 similarly. The internal force vector in the coarse-scale problem is shown below:

$$\mathbf{f}_{\text{int}}^{c,\alpha_E} = \int_{\Omega_{\alpha_E}} (\mathbf{B}_{\alpha_E}^c)^T [\mathbf{P}](\mathbf{X}) dV = \sum_{e \in I_{\alpha_E}} \int_{\Omega_{\alpha_e}} (\mathbf{B}_{\alpha_E}^c)^T [\mathbf{P}] dV, \quad (27)$$

340 where I_{α_E} is index set of fine-scale elements resolving the coarse-scale element, α_E . The
 341 evaluation of each of the element-level entities at the fine scale in the summation given in
 342 Eq. (27) requires the interpolated values of the coarse-scale basis functions and their derivatives
 343 at the integration points of the fine-scale parent domain. This is not readily available as the
 344 coarse-scale functions are defined on the coarse-scale parent domain. To obtain the interpolated
 345 values, a two-scale mapping procedure is employed as discussed in Ref. [29]. This procedure
 346 involves first finding the coordinate of the integration point in the physical domain, using the
 347 fine-scale element isoparametric mapping, and then applying the coarse-scale element inverse
 348 isoparametric mapping to find the coordinates of the same integration point in the coarse-scale
 349 parent domain. The coordinate of the integration point of the fine-scale parent domain (ξ^{f,α_e})
 350 in the coarse-scale parent domain (ξ^{c,α_E}) is given by:

$$\xi^{c,\alpha_E} = \mathcal{M}_c^{-1} \left(\mathcal{M}_f \left(\xi^{f,\alpha_e} \right) \right) \quad (28)$$

351 where \mathcal{M}_f and \mathcal{M}_c denote the fine-scale and coarse-scale isoparametric mappings.

352 3.5 Critical time increment for multiscale and direct numerical 353 simulations

354 For direct numerical simulations of a nonlinear system of governing equations for structural
355 dynamics, linearized stability analysis is performed to determine the critical time increment
356 associated with a time integration method. Firstly, the generalized amplification matrix form
357 is obtained by employing the updates in the time integration scheme to the semi-discrete
358 equations. Then, the generalized amplification eigenvalue problem is decoupled into modal
359 equations by expanding its eigenvectors in terms of the system eigenvectors $\mathbf{K}\Phi = \omega_0^2\mathbf{M}\Phi$
360 [8]. The critical time increment is obtained by restricting the moduli of the complex roots of
361 the characteristic equation for the highest frequency mode in the uncoupled equations to be
362 less than or equal to 1. Noh and Bathe [43] employed the Routh-Hurwitz stability criteria on
363 the characteristic polynomial of the amplification matrix in decoupled modal equations for the
364 explicit sub-step method to obtain a critical time increment as shown below:

$$365 \quad \Delta t_{\text{crit}} = \text{CFL} \max_I \frac{2}{(\omega_0)_I}, \quad (29)$$

366 where I is the index for eigenvalues $(\omega_0)_I$, and the maximum value of Courant–Friedrichs–Lowy
367 (CFL) allowed is $1/p$, where p is the sub-step ratio. It is important to note that the explicit
368 sub-step method [43] has a higher stability limit than the explicit central difference method,
for which the maximum value of CFL allowed is 1.0.

369 In summary, for direct numerical simulations, the critical time increment is obtained in
370 terms of the maximum eigenvalue of the system $\mathbf{K}\Phi = \omega_0^2\mathbf{M}\Phi$. The maximum eigenvalue
371 is estimated based on the element level eigenvalue problem, as the maximum absolute eigen-
372 value of the unconstrained system is upper-bounded by the maximum absolute element level
373 eigenvalue [7]. Moreover, by the Rayleigh nesting theorem, the maximum eigenvalue of the
374 assembled system with essential boundary conditions enforced is bounded by the maximum
375 eigenvalue of the unconstrained element level eigenvalue problem [8]. The estimate of the max-
376 imum eigenvalue for the element level eigenvalue problem in DNS for the examples considered
377 here is discussed in Appendix B.

378 For the multiscale problem, similar ideas are employed to determine the critical time in-
379 crements for both coarse- and fine-scale equations. As the operator-split procedure is used to
380 solve coarse and fine-scale problems iteratively, the stable time increment for both problems
381 can be deduced based on the linearized stability analysis of individual problems.

382 The linearized weak forms of the coarse- and fine-scale PDEs are required for the stability
383 analysis. The discrete form of the linearized weak form for the fine-scale problem is shown in
384 Eq. (23). Following a similar procedure, the discretized form of the linearized weak form for

385 the coarse-scale PDE is obtained as follows:

$$\mathbf{M}^c \tilde{\mathbf{d}} \ddot{\mathbf{d}}^c + \mathbf{K}^c \tilde{\mathbf{d}} \mathbf{d}^c = -\mathbf{M}^c \ddot{\mathbf{d}}^c - \sum_{\alpha=1}^{n_{\text{es}}} \mathbf{M}^{\text{cf}, \alpha} \ddot{\mathbf{d}}^{\text{f}, \alpha} - \mathbf{f}_{\text{int}}^c(\mathbf{d}^c, \mathbf{d}^f) + \mathbf{f}_{\text{ext}}^c, \quad (30)$$

386 where \mathbf{K}^c is the coarse-scale stiffness matrix, and $(\tilde{\mathbf{d}} \mathbf{d}^c, \tilde{\mathbf{d}} \ddot{\mathbf{d}}^c)$ denote the discrete perturbations
387 in coarse-scale fields. \mathbf{K}^c can be obtained using the element-level contributions as shown below:

$$\mathbf{K}^c = \sum_{\alpha, E} (\mathbf{L}_{\alpha E}^c)^T \mathbf{K}_{\alpha E}^c \mathbf{L}_{\alpha E}^c, \quad \text{where} \quad \mathbf{K}_{\alpha E}^c = \sum_{e \in I_{\alpha E}} \int_{\Omega_{\alpha e}} (\mathbf{B}_{\alpha E}^c)^T [\mathbf{D}] \mathbf{B}_{\alpha E}^c dV. \quad (31)$$

388 Following the procedure discussed for DNS, one can obtain the critical time increments for
389 linearized coarse-scale and fine-scale systems given by:

$$\begin{aligned} \text{coarse-scale system-} \quad & \mathbf{K}^c \Phi^c = (\omega_0^c)^2 \mathbf{M}^c \Phi^c, \\ \text{fine-scale system-} \quad & \mathbf{K}^{\text{f}, \alpha} \Phi^{\text{f}, \alpha} = (\omega_0^{\text{f}, \alpha})^2 \mathbf{M}^{\text{f}, \alpha} \Phi^{\text{f}, \alpha}. \end{aligned} \quad (32)$$

390 The critical time increments for the coarse-scale and fine-scale problems are as follows:

$$\begin{aligned} \Delta t_{\text{crit}}^c &= \text{CFL} \max_I \frac{2}{(\omega_0^c)_I}, \\ \Delta t_{\text{crit}}^{\text{f}, \alpha} &= \text{CFL} \max_{\alpha} \left(\max_I \frac{2}{(\omega_0^{\text{f}, \alpha})_I} \right). \end{aligned} \quad (33)$$

391 The maximum eigenvalues for the coarse-scale and fine-scale problems are estimated based
392 on the corresponding element-level problems, similar to DNS. The estimate of the maximum
393 eigenvalue for the element-level eigenvalue problem in coarse and fine-scale differential equa-
394 tions in VME simulations for the examples considered here is discussed in Appendix C. It is
395 expected that the stable time increment of the coarse-scale problem will be larger than that
396 of the fine-scale problem, as $\omega_0^{\text{f}, \alpha} \gg \omega_0^c$. Hence, the stable time increment for the multiscale
397 problems based on the integration schemes can be deduced as follows:

- 398 • For EE-CDM or EE-SSM algorithm, $\Delta t_{\text{crit}} = \Delta t_{\text{crit}}^{\text{f}, \alpha}$. The CFL is chosen based on the
399 integration scheme employed.
- 400 • For the EI-SSM algorithm, it is given by $\Delta t_{\text{crit}} = \Delta t_{\text{crit}}^c$, as the implicit update for the
401 fine-scale problem is unconditionally stable.

4 Numerical Verification

402 This section presents numerical examples of wave propagation in a one-dimensional domain
403 modeled using the compressible Neo-Hookean material model (see Appendix A). While the

numerical schemes developed in this manuscript apply to multidimensional problems, one-dimensional cases directly address the issues associated with multiscale time integration stability. We therefore restricted the numerical analysis to one-dimensional cases. First, the performance of time integration methods, including the explicit central difference method, explicit-explicit sub-step method, and explicit-implicit sub-step method, is evaluated for wave propagation in VME simulations for a homogeneous microstructure. Next, the effect of contrast in the elastic modulus of a heterogeneous microstructure on wave propagation is examined. Then, the effect of different initial displacement fields on wave propagation is examined in multiscale simulations, demonstrating that the VME approach accurately reproduces key wave propagation characteristics such as dispersion and attenuation, consistent with DNS results. Finally, the computational performance of explicit-explicit and explicit-implicit time integration schemes is assessed under varying contrasts in the elastic modulus of a heterogeneous microstructure.

The original equation of motion in the 1-D case is expressed in the following form:

$$\frac{\partial P(X, t)}{\partial X} = \rho_0(X) \frac{\partial^2 u(X, t)}{\partial t^2}, \quad (34)$$

where P denotes the first Piola-Kirchhoff stress, and u is the displacement. The non-dimensional form of Eq. (34) is obtained by introducing the following entities:

$$\tilde{X} = \frac{X}{L}; \quad \tilde{t} = \frac{v t}{L}; \quad \tilde{P} = \frac{P}{E^A}; \quad \tilde{u} = \frac{u}{L}, \quad (35)$$

where L is the length of the domain, $v = \sqrt{E^A/\rho_0^A}$ is the wave speed of material A , E^A is the Young's modulus of material A , and ρ_0^A is the mass density of material A in the reference configuration. Substituting Eq. (35) into Eq. (34), we obtain the non-dimensional form of the equation of motion:

$$\frac{\partial \tilde{P}(\tilde{X}, \tilde{t})}{\partial \tilde{X}} = \frac{\rho_0(\tilde{X})}{\rho_0^A} \frac{\partial^2 \tilde{u}(\tilde{X}, \tilde{t})}{\partial \tilde{t}^2}. \quad (36)$$

The corresponding multiscale system is derived as described in Section 2. In what follows, the non-dimensional form of the governing equation is solved for all numerical examples demonstrated below. The (\cdot) symbol is omitted from the non-dimensional entities for simplicity of the presentation.

The initial conditions considered for all the examples demonstrated below are as follows ($X \in [-1/2, 1/2]$):

$$u(X, 0) = a \left(1 - \tanh^2 \left(\frac{X}{c} \right) \right), \quad (37)$$

$$\dot{u}(X, 0) = 0,$$

431 and, the boundary conditions are $u(-1/2, t) = 0$ and $u(1/2, t) = 0$, unless stated otherwise.

432 For all multiscale simulations (referred to as VME below) reported in this work, unless
 433 otherwise mentioned, the 1-D domain is discretized using $n_{es} = 100$ unit cells, $n_{ecp} = 1$ coarse
 434 element per unit cell, and $n_{ef} = 8$ fine-scale quadratic elements for each unit cell. The results
 435 of the multiscale simulations are compared with the direct numerical simulations (DNS), which
 436 are obtained using the finite element method, where the material microstructure is resolved
 437 throughout the problem domain. To be consistent, the domain is discretized into 800 quadratic
 438 elements for DNS. Further refinement of the domain does not result in significant accuracy
 439 improvements in the cases discussed below. The sub-step ratio for the EE-SSM or EI-SSM of
 440 time integration in VME simulations and explicit sub-step integration in DNS is taken to be
 441 $p = 0.54$, as suggested in Ref. [43]. The tolerance value for convergence between coarse and
 442 fine-scale problems in VME simulations is 1E-3 for all time-integration methods. The tolerance
 443 value for the Newton-Raphson iterations in the fine-scale problem for implicit updates at both
 444 sub-step and full step is 1E-10.

445 4.1 Homogeneous domain

446 In this section, we assess the performance of different time integration schemes for wave prop-
 447 agation for both VME and DNS simulations. The initial displacement profile is obtained using
 448 Eq. (37) with $a = 0.04$ and $c = 0.05$.

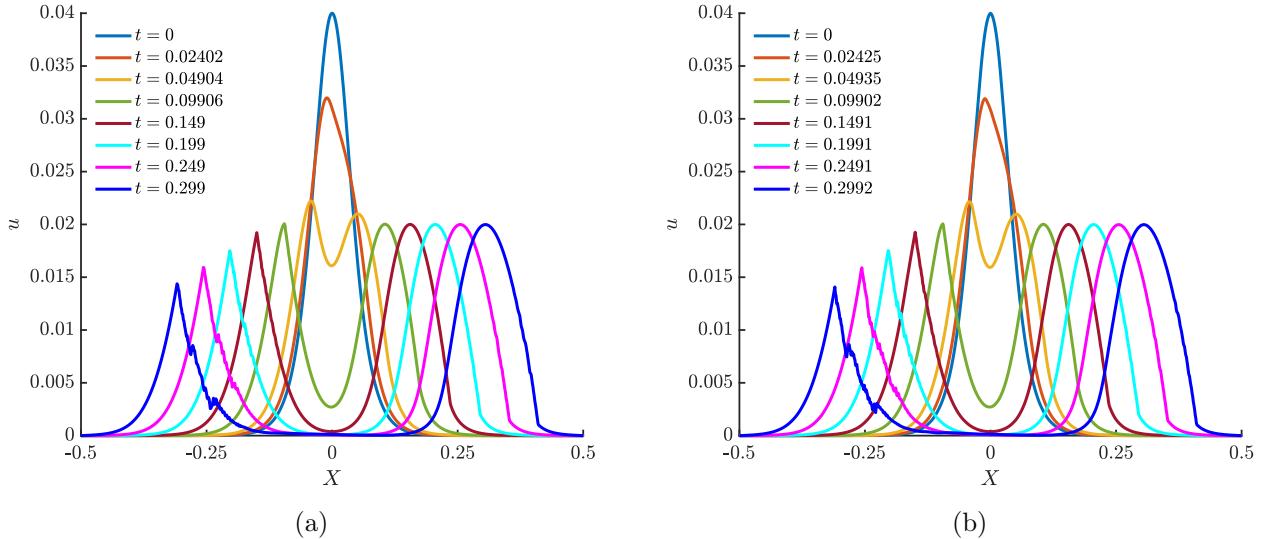


Figure 1: Evolution of displacement with time predicted using the explicit central difference methods for the homogeneous material case with (a) DNS and (b) VME methods.

449 We first consider the homogeneous microstructure within all coarse-scale elements, by which
 450 $\rho_0/\rho_0^A = 1$ and $E/E_A = 1$. Figure 1 shows the evolution of the total displacement profile for

a homogeneous microstructure using the explicit central difference method for both DNS and VME (using the EE-CDM approach) simulations. The results of the VME simulations are reported in the total form where the fine- and coarse-scale parts of the solution are evaluated separately and summed. Both the DNS and VME simulation results show that the initial displacement profile induces two waves traveling in opposite directions, consistent with the D'Alembert solution for linear wave propagation. The evolution of (element-averaged) stretch along the X direction given by $F = 1 + du/dX$, is shown in Fig. 2. Material points with $F > 1$ indicate stretching, while those with $F < 1$ indicate compression.

Due to the geometric and material nonlinearities in the Neo-Hookean model, the local wave speed becomes amplitude-dependent, leading to wave steepening. Specifically, the wave speed is lower in the tensile region than in the compressive region as the tangent modulus in compression is larger than in tension, for the same magnitude of displacement gradient (see Eq. (B.3) in Appendix B), resulting in asymmetric propagation. As shown in Fig. 1, this asymmetry manifests as a narrower crest on the tensile (left) side and a broader crest on the compressive (right) side, along with a reduction in amplitude on the tensile side. The compressive part of the traveling wave on the left side travels faster than the tensile part, leading to a reduction of the peak of the displacement wave as a function of time. Spurious oscillations appear in the displacement and strain wave profiles due to numerical dispersion errors introduced by the central difference method, particularly affecting high-frequency components of the solution. The numerical dispersion effects are apparent after $t = 0.249$ for both DNS and VME simulations.

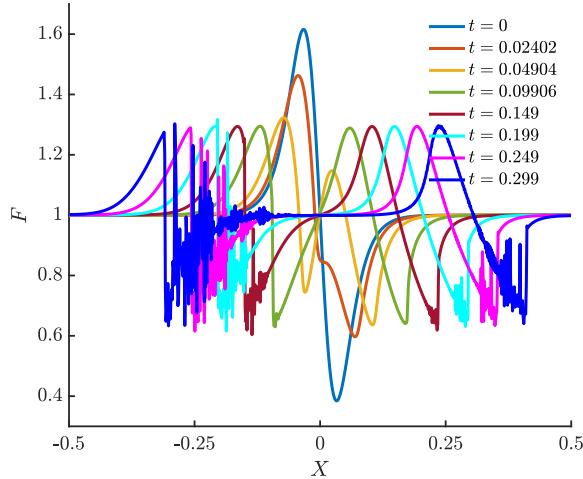


Figure 2: Averaged stretch over the element for the homogeneous material case in DNS.

Figure 3 shows a comparison of the displacement waves predicted by the VME and DNS approaches using dissipative integration schemes. The DNS simulation employs the explicit

474 sub-step integration method proposed in Ref. [43], whereas for VME simulations, the EE-SSM
 475 and EI-SSM are utilized. $CFL = 1$ is used for both DNS and EE-SSM-based VME simulations,
 476 while $CFL = 0.5$ is taken for EI-SSM-based VME simulations. As evident in Fig. 3, spurious
 477 oscillations do not develop when the dissipative methods are employed. This is because the
 478 contribution of high-frequency modes to the overall solution is reduced in these methods by
 479 decreasing the spectral radius of the amplification matrix for shorter wavelengths [5, 43]. In
 480 Fig. 3, the relative error based on the L^∞ norm of the total displacement field between the VME
 481 and DNS simulations is 0.0051 for EE-SSM at $t = 0.2993$, and 0.0259 for EI-SSM at $t = 0.2998$.
 482 Hence, for the same discretization, the EE-SSM and EI-SSM-based VME simulations remove
 483 the high-frequency oscillations that are present for the CDM-based VME simulations. We note
 484 that, unlike computational homogenization-based methods, the fine-scale solution in VME is
 485 not necessarily induced by heterogeneity in the microstructure. In the case of a homogeneous
 486 domain, the fine-scale solution captures the discretization errors induced by the coarse-scale
 487 grid and effectively improves the accuracy of the solution. Figure 3a includes the results
 488 obtained using the coarse grid approximation alone, which deviates from the multiscale and
 489 the DNS solutions, especially at later times, where the relative error of the total displacement
 490 compared to DNS is 0.1504 at $t = 0.2993$. This is due to the accumulation of large numerical
 491 dispersion errors with coarse-grid approximation alone in the VME simulation.

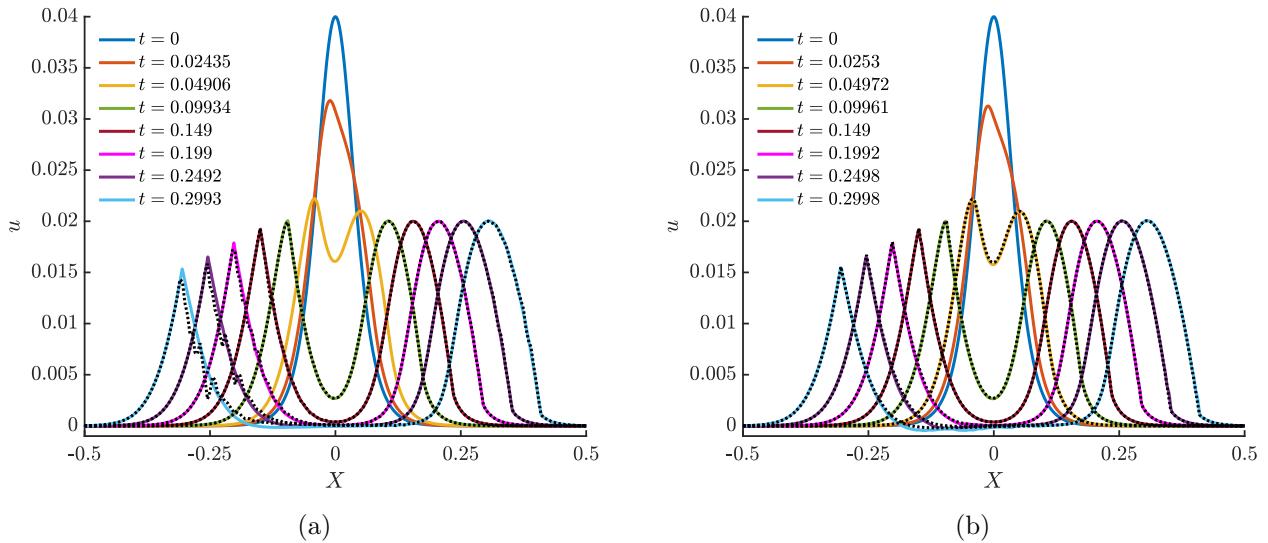


Figure 3: Evolution of displacement profile with time for homogeneous microstructures using (a) EE-SSM, (b) EI-SSM in VME simulations. In both figures, the colored solid lines correspond to VME results, while the dotted lines in (a) correspond to the EE-SSM-based VME with coarse-grid approximation alone, and in (b) correspond to DNS, at the same non-dimensional times as the colored solid lines.

4.2 Heterogeneous domain

In this section, we study the effect of contrast in the elastic modulus in a heterogeneous microstructure on wave propagation. Let the domain consist of a repeated two-phase microstructure such that:

$$E = \begin{cases} E^A, & \text{if } X \in \Omega_A; \\ C E^A, & \text{if } X \in \Omega_B; \end{cases} \quad (38)$$

where C is the modulus contrast ratio, $\Omega_A = [-kl - L/2, (k + \beta)l - L/2]$ and $\Omega_B = [(k + \beta)l - L/2, (k + 1)l - L/2]$, with l the size of the microstructure, β the fraction of material A , and $k \in \{0, 1, \dots, (L/l) - 1\}$. The impedance mismatch is taken to be generated by the modulus contrast alone, and the mass density in the reference configuration is taken as constant, i.e. $\rho_0 = \rho_0^A$.

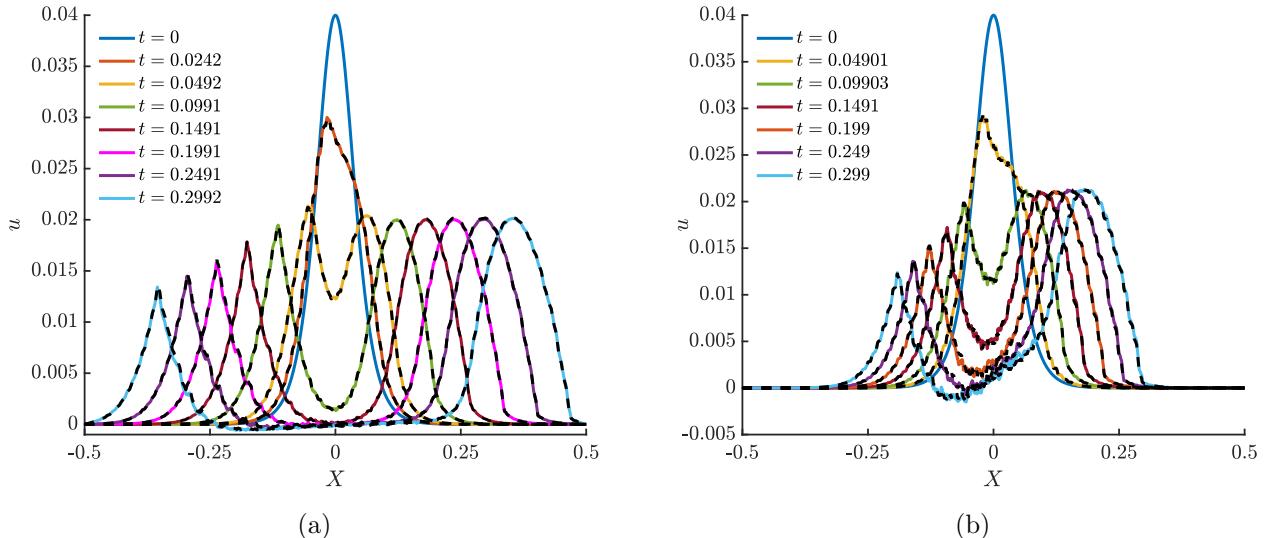


Figure 4: Evolution of displacement profile with time using EE-SSM for VME and explicit sub-step method for DNS simulations of heterogeneous microstructures (a) $C = 2$, (b) $C = 0.2$. In both figures, the colored solid lines correspond to VME results, and the black dashed lines correspond to the DNS results.

Figures 4a-b show the propagation of the displacement wave induced by the initial displacement profile obtained using Eq. 37 with $a = 0.04$ and $c = 0.05$ for $C = 2$ (i.e., phase contrast of 2) and $C = 0.2$ (phase contrast of 5), respectively. The microstructure parameters are set to $l = 0.01$, $\beta = 0.5$, and each unit cell is associated with one coarse-scale element i.e., $n_{\text{eep}} = 1$. The DNS simulations were performed using the explicit sub-step integration method, whereas the VME simulations employ EE-SSM for time integration. A CFL number of 1 is used for $C = 2$, while CFL = 0.2 is used for $C = 0.2$. The CFL was reduced for the $C = 0.2$ case, as it was observed that using larger values of CFL led to numerical instability. This can happen in nonlinear problems, as the stable time increment is obtained from linearized stability

analysis, and the actual response may deviate from the linear approximation. As observed in Fig. 4, the displacement waves travel faster or slower depending on the increase/decrease in the elastic modulus compared to the homogeneous microstructure. Due to contrast in the elastic modulus, especially for $C = 0.2$, wave reflections are more prominent, leading to oscillations in the wave profile with evolution in time. However, no significant wave dispersion arises from the microstructural heterogeneity, as the initial wavelength is substantially larger than the microstructural length scale. As a result, the propagating wave interaction with the microstructure is limited. The VME simulation captures the resulting oscillations with reasonable accuracy compared to the DNS results, with the relative error in the total displacement compared to DNS being 0.02496 for $C = 2$, and 0.1218 for $C = 0.2$, both at $t = 0.299$.

Figure 5 shows the evolution of the displacement field for a heterogeneous microstructure with $C = 0.01$ (phase contrast of 100) using EE-SSM for VME simulations and explicit sub-step method for DNS. The initial displacement profile is chosen with $a = 0.005$ in Eq. (37), and this is because if larger amplitudes are taken, then it results in unphysical compressive strains with stretch, $F < 0.01$. The CFL = 0.1 is taken for both VME and DNS results. The relative error in the displacement field obtained from VME compared to DNS is 0.27138 at $t = 0.299$, and this is resulting from a relatively larger tolerance chosen for convergence in iterations between coarse- and fine-scale problems.

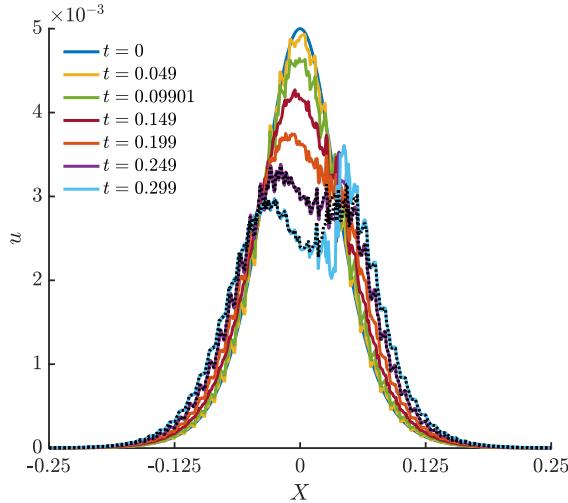


Figure 5: Evolution of displacement profile with time using EE-SSM for VME simulations for a heterogeneous microstructure with $C = 0.01$ and CFL = 0.1. The dashed black lines show DNS results at the same non-dimensional times using the explicit sub-step method.

To investigate wave dispersion effects, the size of the microstructure is increased, thereby promoting the interaction between the propagating wave and the material heterogeneity. We consider $l = 0.04$, i.e., $n_{\text{es}} = 25$ microstructures in the domain, and use a fine-scale grid $n_{\text{ef}} = 32$ per unit cell, and three coarse-scale grids are considered: (a) $n_{\text{ecp}} = 1$, (b) $n_{\text{ecp}} = 2$,

532 and (c) $n_{\text{ecp}} = 4$. The error in the wave propagation is evaluated as a function of the number
 533 of elements in a patch of coarse-scale elements that discretize a unit cell. Figure 6a shows
 534 the evolution of the displacement profiles for smaller ($l = 0.01$, $n_{\text{ecp}} = 1$) and larger ($l =$
 535 0.04, $n_{\text{ecp}} = 4$) microstructures. The VME simulations employ the EE-SSM time integration
 536 scheme with a CFL number of 0.5. When the microstructure is larger, the displacement wave
 537 has a stronger interaction with the microstructure, resulting in wave dispersion. When the
 538 microstructural size increases, the propagating displacement wave interacts more strongly with
 539 the microstructure, and dispersive effects become significant. For example, the left-traveling
 540 peak is smaller for $l = 0.04$ compared to $l = 0.01$ – a consequence of dispersion. Similarly, on
 541 the right side, the broadening of the wave profile further confirms the presence of dispersive
 542 behavior. Figure 6b shows the relative error for $l = 0.04$ compared to DNS for an increasing
 543 number of elements in a patch of coarse elements, i.e., n_{ecp} . The DNS were obtained with
 544 $n_{\text{el}} = 800$ elements with a CFL = 0.5. As expected, the relative errors decrease with an
 545 increase in n_{ecp} and also increase with time. This will become even more critical as the
 546 microstructural size increases, particularly in higher-dimensional problems.

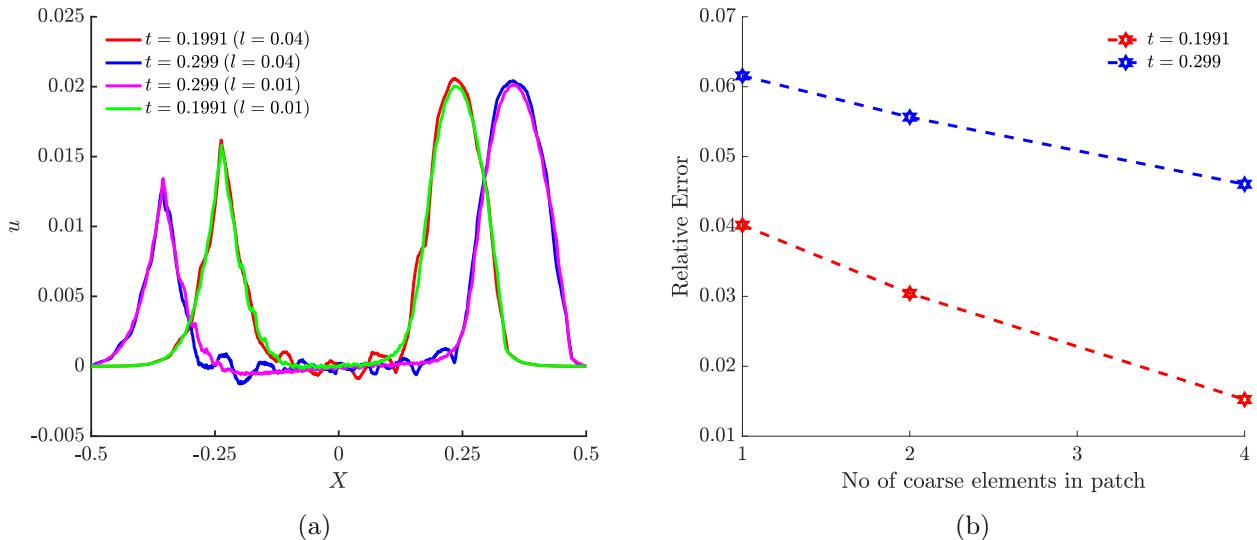


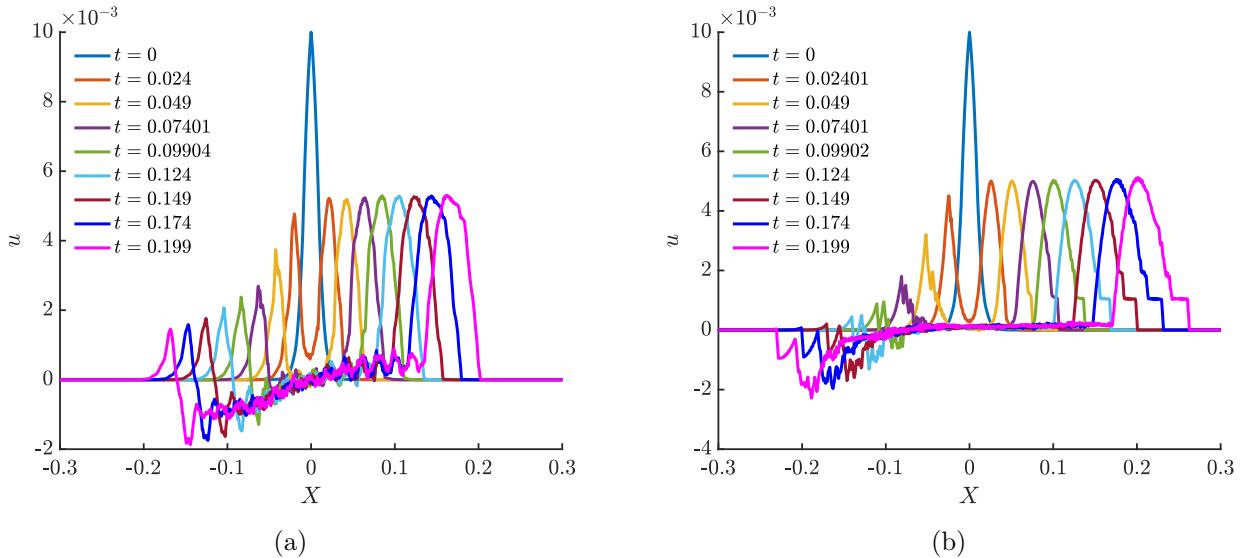
Figure 6: Wave dispersion using EE-SSM for VME simulations of heterogeneous microstructure with $C = 2$, $n_{\text{es}} = 25$, and $n_{\text{ef}} = 32$. (a) $n_{\text{ecp}} = 4$ for $l = 0.04$, and $n_{\text{ecp}} = 1$ for $l = 0.01$, (b) comparison of relative error with respect to DNS as a function of n_{ecp} for $l = 0.04$.

4.3 Effect of initial displacement conditions

In this section, we examine how the initial displacement profile influences wave propagation. In particular, we focus on the role of wavelength, controlled by the parameter c in Eq. (37), for a heterogeneous microstructure. The effect of amplitude on the solution has already been

551 demonstrated through its deviation from the D'Alembert solution as discussed in Section 4.1.
 552 For the heterogeneous case, the microstructure is defined with a phase contrast of 2 (by setting
 553 $C = 0.5$), size $l = 0.01$, and the coarse-scale mesh with $n_{\text{ecp}} = 1$. The initial displacement field
 554 is specified with $a = 0.01$, $c = 0.01$ in Eq. (37). To distinguish the roles of nonlinearity and
 555 heterogeneity, we also compare the results with those of the homogeneous problem.

556 Figure 7 shows the results obtained from the VME simulations for both heterogeneous
 557 and homogeneous problems. The relative error in the displacement field compared to DNS at
 558 $t = 0.199$ is 0.0976 for the heterogeneous case, and 0.2226 for the homogeneous case. Since
 559 the initial wavelength is small, it is observed that the propagating wave interacts with the het-
 560 erogeneous microstructure, resulting in oscillations as evident in Fig. 7a. In the homogeneous
 561 case, the oscillations are relatively small and are primarily caused by the variation in the local
 562 wave speed, unlike the heterogeneous problem.



563 Figure 7: Effect of initial displacement profile on wave propagation using $a = 0.01$, $c = 0.01$ for (a)
 564 heterogeneous case ($C = 0.5$), and (b) homogeneous case ($C = 1.0$), in VME simulations.

4.4 Computational cost comparison for integration schemes

564 We compare the computation time for different time integration schemes in VME for different
 565 contrast values in the elastic modulus for a heterogeneous microstructure. Table 1 reports the
 566 run time in seconds and the relative error with respect to DNS for various contrast values
 567 in elastic modulus and corresponding initial displacement condition (by setting a and c in
 568 Eq. (37)) up to non-dimensional time $t = 0.2$ using EE-SSM and EI-SSM integration schemes.
 569 The size of the microstructure in all cases is $l = 0.01$, and each unit cell is mapped to a single
 570 coarse-scale element i.e., $n_{\text{ecp}} = 1$. The CFL values are chosen to obtain stable evolution for

571 the multiscale problem, and the relative errors are reported for different CFL, specifically for
 572 EI-SSM-based time integration to demonstrate the relative gain in the accuracy at the expense
 573 of compute time.

574 The reported computational times are obtained from a single-processor implementation
 575 and are higher than those of DNS, since the multiscale coupled problems are solved iteratively
 576 at each time increment. The formulation, however, naturally lends itself to parallelization,
 577 as the uncoupled fine-scale problems for individual unit cells can be computed independently,
 578 thereby enabling a reduction in overall runtime. Without model order reduction, the VME
 579 approach involves essentially the same number of degrees of freedom as DNS. Nevertheless, its
 580 structure provides a foundation for more efficient methods in which the fine-scale problem can
 581 be replaced by a data-driven surrogate model.

Table 1: Comparison of computational cost and relative error with respect to DNS of different time integration methods for varying contrasts in elastic modulus and different CFL values.

Contrast	Initial displacement	Integration method	n_{ef}	CFL	Compute time (secs)	Relative error
$C = 1.0$	$a = 0.04, c = 0.05$	EE-SSM	8	1.0	264.03	0.00187
		EI-SSM		1.0	148.35	0.0242
		EI-SSM		0.5	487.07	0.0168
$C = 0.5$	$a = 0.04, c = 0.05$	EE-SSM	8	1.0	289.49	0.0104
		EI-SSM		1.0	174.02	0.0806
		EI-SSM		0.5	550.82	0.0588
		EI-SSM		0.25	1726.32	0.0557
$C = 0.2$	$a = 0.04, c = 0.05$	EE-SSM	8	0.2	1549.77	0.07801
		EI-SSM		0.5	544.48	0.3467
		EI-SSM		0.1	5171.20	0.2014
$C = 0.01$	$a = 0.005, c = 0.05$	EE-SSM	8	0.1	1059.19	0.0415
		EI-SSM		0.05	6515.87	0.2166

582 As shown in Table 1, for the homogeneous case, the EI-SSM scheme is less expensive than
 583 EE-SSM while maintaining a relative error within 2.5% for EI-SSM. The lower cost of EI-SSM
 584 arises because the stable time increment is controlled by the coarse-scale problem, allowing
 585 for larger time steps that offset the additional expense of solving nonlinear systems at the fine
 586 scale. For the heterogeneous case with moderate stiffness contrast ($C = 0.5$), EI-SSM still
 587 permits large stable time increments; however, achieving a relative error within 6% compared
 588 to DNS requires smaller time steps. For higher contrasts, $C = 0.2$ and $C = 0.01$, EI-SSM
 589 yields relatively large errors, and EE-SSM outperforms it in both accuracy and computational
 590 cost. In fact, even with fewer fine-scale elements, EE-SSM achieves higher accuracy.

591 Overall, while the implicit scheme benefits from larger stability limits, it suffers from re-
 592 duced accuracy. This is because the difference between the stable time increment for explicit

593 evolution of the fine-scale problem and the time increment required for accurate evolution is
594 small, given the second-order accuracy of the current scheme [43, 6, 5]. This limitation could
595 be alleviated by adopting higher-order integration schemes, which would allow larger implicit
596 time steps while retaining the required accuracy for the multiscale problem. Another way
597 to improve the accuracy of the EI-SSM scheme is to impose a tighter convergence tolerance
598 between the fine- and coarse-scale problems within the operator-split procedure. However, this
599 would also increase the overall computational cost.

600 5 Conclusion

601 The proposed multiscale computational framework enables the simulation of wave propagation
602 under scale-inseparable conditions, including short-wavelength regimes, while accounting for
603 both material and geometric nonlinearities. By employing an additive decomposition of the
604 solution fields into coarse- and fine-scale components, a coupled two-scale system of equations
605 is derived. The framework allows the discretization of each unit cell with a patch of coarse-
606 scale elements, which is essential to accurately capture wave propagation, especially in short-
607 wavelength regimes. The coarse-scale semi-discrete equations are integrated explicitly, while
608 the fine-scale equations are integrated either explicitly or implicitly, using both dissipative and
609 non-dissipative time integration schemes.

610 The numerical examples demonstrate that the VME method accurately captures wave
611 dispersion, attenuation, and wave steepening arising from microstructural heterogeneity, mi-
612 crostructural size, and nonlinearities in the constitutive model, with results in close agreement
613 with DNS. It is shown how the relative error reduces using multiple coarse-scale elements that
614 discretize a unit cell, when the microstructure size is large, leading to wave dispersion due to
615 heterogeneity. It is also observed that the EE-SSM scheme is computationally less expensive
616 than EI-SSM for heterogeneous one-dimensional problems. Although EI-SSM permits larger
617 stability limits, its accuracy is limited by the second-order nature of the integration schemes.
618 In higher-dimensional heterogeneous problems, however, the explicit-implicit approach may
619 provide computational advantages, depending on the relative constraints imposed by stability
620 and accuracy requirements for the fine-scale problem.

621 Future work will focus on implementing the current framework in higher dimensions for
622 modeling the dynamic response of architected materials. An additional key direction is the
623 development of reduced-order VME models using data-driven surrogate approaches, where the
624 fine-scale problem is replaced with a trained surrogate model, thereby achieving computational
625 efficiency beyond DNS.

626 Acknowledgments

627 The authors wish to acknowledge the financial support of the Army Research Office through
 628 the Solid Mechanics Program (Grant No. W911NF2320134; Program Officer: Dr. Denise
 629 Ford).

630 A Neo-Hookean material

631 The one-dimensional energy density function for the Neo-Hookean material model is obtained
 632 from the multi-dimensional form shown below:

$$\psi(\mathbf{F}) = \frac{\mu}{2} \left[\text{tr}(\mathbf{F}^T \cdot \mathbf{F}) - n_{\text{dim}} \right] - \mu \ln(J) + \frac{\lambda}{2} \ln^2(J), \quad (\text{A.1})$$

633 where J denotes the Jacobian of deformation gradient tensor $J = \det(\mathbf{F})$, μ and λ are Lame's
 634 parameters, and n_{dim} denotes the number of spatial dimensions. The first Piola-Kirchhoff
 635 stress for the multi-dimensional case is given below:

$$\mathbf{P} = \frac{\partial \psi(\mathbf{F})}{\partial \mathbf{F}} = \left[\lambda \ln(J) - \mu \right] \mathbf{F}^{-T} + \mu \mathbf{F}. \quad (\text{A.2})$$

636 In the one-dimensional case, the deformation gradient tensor can be written as:

$$\mathbf{F} = \left(1 + \frac{\partial u}{\partial X} \right) \mathbf{e}_1 \otimes \mathbf{e}_1 + \mathbf{e}_2 \otimes \mathbf{e}_2 + \mathbf{e}_3 \otimes \mathbf{e}_3. \quad (\text{A.3})$$

637 Furthermore, assuming Poisson's ratio to be $\nu = 0$, yields the Lamé parameters in the model
 638 to be $\lambda = 0$ and $\mu = E/2$. Denoting $F := 1 + \frac{\partial u}{\partial X}$, the energy density function and the only
 639 non-zero component of the first Piola-Kirchhoff stress tensor, denoted as P , in the 1-D case
 640 are obtained as given below:

$$\begin{aligned} \psi(F) &= \frac{E}{4} \left(\left(1 + \frac{\partial u}{\partial X} \right)^2 - 1 - 2 \ln \left(1 + \frac{\partial u}{\partial X} \right) \right), \\ P &= \frac{\partial \psi}{\partial F} \equiv \frac{E}{2} \left(1 + \frac{\partial u}{\partial X} - \left(1 + \frac{\partial u}{\partial X} \right)^{-1} \right). \end{aligned} \quad (\text{A.4})$$

641 B Estimate of critical time step in 1-D: DNS

642 The element-level eigenvalue problem for the DNS is obtained by taking the variation of the
 643 weak form of the governing equations given in Eq. (5). In the 1-D case, the first variation
 644 along the direction $(\tilde{du}, \tilde{d\ddot{u}})$ of the weak form with the Neo-Hookean material (ignoring the

645 forcing terms), is obtained as given below:

$$\int_{\Omega} \delta u \rho_0 \frac{\partial^2(\tilde{d}u)}{\partial t^2} dX + \int_{\Omega} \frac{\partial \delta u}{\partial X} \frac{E}{2} \left(1 + \frac{1}{(\partial x / \partial X)^2} \right) \frac{\partial \tilde{d}u}{\partial X} dX = 0, \quad (B.1)$$

646 where the current displacement field is u , and the current deformed configuration is $x = X + u$.

647 The element-level mass matrix and linearized stiffness matrix are given by:

$$\begin{aligned} \mathbf{K}_e &= \int_{\Omega_e} \frac{E}{2} \left(1 + \frac{1}{(\partial x / \partial X)^2} \right) (\mathbf{B}_e)^T \mathbf{B}_e dX, \\ \mathbf{M}_e &= \int_{\Omega_e} \rho_0 (\mathbf{N}_e)^T \mathbf{N}_e dX, \end{aligned} \quad (B.2)$$

648 where \mathbf{N}_e and \mathbf{B}_e are the shape function matrix and shape function gradient matrix for an ele-
 649 ment Ω_e . The maximum eigenvalue of the element-level problem is bounded by the maximum
 650 eigenvalue of the integrands of stiffness and mass matrices at each quadrature point [7, 8].
 651 For a quadratic element with a diagonal mass matrix by row-sum lumping, the maximum
 652 element-level eigenvalue is estimated to be [8]:

$$(\omega_0)_e = \frac{2\sqrt{6}}{h_e} \max_{\zeta_{Q,e}} \left(\sqrt{\frac{E}{2\rho_0} \left(1 + \frac{1}{(\partial x / \partial X)^2} \right)} \right), \quad (B.3)$$

653 where $\max(\cdot)$ denotes the maximum over all the quadrature points in an element ($\zeta_{Q,e}$), and
 654 h_e denotes the length of an element. The critical time step is then obtained by taking the
 655 maximum eigenvalue over all elements as given below:

$$\Delta t_{\text{crit}} = \text{CFL} \max_e \frac{2}{(\omega_0)_e} = \text{CFL} \min_e \left(\frac{h_e}{\sqrt{6} \zeta_{Q,e}} \left(\frac{1}{\sqrt{\frac{E}{2\rho_0} \left(1 + \frac{1}{(\partial x / \partial X)^2} \right)}} \right) \right). \quad (B.4)$$

656 As noted in [8], these estimates are good for problems with C^1 constitutive laws and smooth
 657 response, and for rough problems like impact, reductions in time steps are advised by appro-
 658 priately choosing smaller values of CFL.

659 C Estimate of critical time step in 1-D: multiscale 660 problem

661 The critical element-level eigenvalue problems for the coarse and fine-scale problems in VME
 662 are obtained by taking the first variation of the weak form of the corresponding equations.
 663 Following the procedure discussed in Appendix B, one can obtain maximum element-level

664 eigenvalues for fine and coarse-scale elements for a 1-D problem as given below:

$$\begin{aligned} (\omega_0^{f_\alpha})_e &= \frac{2\sqrt{6}}{h_e^{f_\alpha}} \max_{\zeta_{Q,e}} \left(\sqrt{\frac{E}{2\rho_0} \left(1 + \frac{1}{(\partial x/\partial X)^2} \right)} \right), \\ (\omega_0^c)_{\alpha,E} &= \frac{2\sqrt{6}}{h_{\alpha,E}^c} \max_{\zeta_{Q,\alpha,E}} \left(\sqrt{\frac{E}{2\rho_0} \left(1 + \frac{1}{(\partial x/\partial X)^2} \right)} \right), \end{aligned} \quad (C.1)$$

665 where $(\omega_0^{f_\alpha})_e$ and $(\omega_0^c)_{\alpha,E}$ are the fine and coarse-scale maximum eigenvalues for the corre-
 666 sponding fine-scale element (e) and coarse-scale element (α, E). $h_e^{f_\alpha}$ and $h_{\alpha,E}^c$ are the element
 667 lengths for fine and coarse-scale elements, respectively. It must be noted that, for the coarse-
 668 scale estimate above, the maximum is defined over all the quadrature points in all fine-scale
 669 elements (e) within a coarse-scale element (α, E). Finally, the critical time steps for fine and
 670 coarse-scale problems are given by:

$$\begin{aligned} \Delta t_{\text{crit}}^{f_\alpha} &= \text{CFL} \max_e \frac{2}{(\omega_0^{f_\alpha})_e} = \text{CFL} \min_e \left(\frac{h_e^{f_\alpha}}{\sqrt{6}} \min_{\zeta_{Q,e}} \left(\frac{1}{\sqrt{\frac{E}{2\rho_0} \left(1 + \frac{1}{(\partial x/\partial X)^2} \right)}} \right) \right), \\ \Delta t_{\text{crit}}^c &= \text{CFL} \max_{\alpha,E} \frac{2}{(\omega_0^c)_{\alpha,E}} = \text{CFL} \min_{\alpha,E} \left(\frac{h_{\alpha,E}^c}{\sqrt{6}} \min_{\zeta_{Q,\alpha,E}} \left(\frac{1}{\sqrt{\frac{E}{2\rho_0} \left(1 + \frac{1}{(\partial x/\partial X)^2} \right)}} \right) \right). \end{aligned} \quad (C.2)$$

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